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Multiple shots averaging in Laser Flash 1 measurement 2

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12 Abstract: The laser flash method is a well-known procedure to determine the thermal 13 diffusivity of a wide range of materials. However, in some cases there is the need of limiting 14 the input power, or measuring materials with high thermal capacity, or investigating thick samples. These conditions lead to a reduction of the signal-to-noise ratio. Therefore, we 15 propose a new laser flash control and data acquisition system, that is able to repeat multiple 16 17 times the emission of the laser impulse and the measurement of the thermal response of the 18 specimen. With the average of several measurements, it is possible to obtain a decrease of the 19 noise when working with low power inputs.

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21 1. Introduction

22 The nondestructive evaluation of materials is a fundamental tool for the industrial production, 23 as it can significantly contribute to improve the design of the components and to the 24 assessment of the production quality [1,2]. Several methods and techniques are available to analyze material samples and investigate the chemical, structural, or thermal properties [3]. 25 The latter are important, for example during the design phase of a component. The thermal 26 27 properties are used as input of simulation models [4] and providing accurate data to the 28 models is the basic condition to obtain reliable predictions. A predetermined value of a 29 thermal property could be also the desired requirement of an industrial component. The 30 measurement on a component coming from different production batches could help the 31 manufacturer to identify anomalous products [5]. Broadly speaking, the use of a 32 nondestructive method is a great advantage, especially when dealing with real-size 33 mockups [6] or when performing on-site evaluations [7].

Among all available methods for determining the thermal properties [8], one of the most 34 35 popular is the Laser Flash Method (LFM) [9]. In the standard LFM procedure where a heat 36 flux is applied on the front face of a specimen and the temperature is measured on the back of 37 the specimen [10-12]. Single-side configurations are available [13]. The LFM is typically 38 applied to small size disk-shaped specimens in order to measure their thermal diffusivity (α value). The main advantages of the method are its simplicity, speed of measurement, and the 39 40 possibility to measure the thermal diffusivity of a large variety of materials within a wide 41 temperature range. The LFM is chosen also because it gives the possibility of measuring the 42 thermal conductivity. To this end a procedure with adequate reference samples must be set 43 up. In alternative, knowledge of the density (ρ) , that can be measured through Archimedes principle, and of the specific heat (c_n), that can be obtained through differential scanning 44 45 calorimetry (DSC), makes it possible to derive the thermal conductivity as the product of the 46 density (ρ), the specific heat (c_p), and the thermal diffusivity (α) [14].

1 The original LFM has been widely studied and applied, expecially for the characterization 2 of thermal barrier coatings (TBC) [15-18]. Several modifications have been proposed, both 3 from the mathematical [19] and from the experimental standpoint [20,21] to improve the 4 results and quantify the possibilities and limitations of the method. One critical issue is 5 limiting the input power, that for some ranges of thermal properties and thickness of the 6 specimen may lead to an unwanted and nonuniform overheating [17]. Vozàr and 7 Hohenauer [22] proposed to divide the energy of a single laser pulse into smaller repeated 8 pulses, obtaining results comparable to the traditional technique. Recently, Ruffio et al. [23] 9 explored the use of a high speed laser pulse train in order to improve the signal-to-noise ratio 10 (SNR). This study proposes a novel experimental setup that allows the automatic repetition of single pulses with a user defined time delay between each pulse. This leads to an increase of 11 12 the SNR, as shown in the following sections. This allows to use the LFM method with lower 13 input power on specimens that have an unfavorable combination of thermal properties and 14 thickness.

15 2. Thermal diffusivity measurement – Laser Flash Method

16 2.1 Mathematical model

17 Several mathematical models are available to describe the heat conduction problem of this 18 method. Considering the specimen under analysis as a slab of thickness l[m], the heating 19 pulse can be modeled as a Dirac delta function $\delta(t)$. The heating is uniformly distributed over 20 the slab surface and produces a one dimensional thermal diffusion through the thickness of 21 the specimen. The heat transfer problem is described by Eq. (1):

$$u_{zz} - \frac{1}{\alpha}u_t = 0 \tag{1}$$

22

23 where u=u(z,t) is the function describing the temperature in the body, that depends on the 24 space coordinate z and time t[s], and α is the thermal diffusivity $[m^2 s^{-1}]$. It is possible to 25 choose two different hypotheses: the adiabatic case and the non-adiabatic case.

26 Under the adiabatic hypothesis, the exchange with the environment (*h* is the heat exchange coefficient [W m⁻² K⁻¹]) is neglected, as shown in Fig. 1.



33 The boundary conditions are:

$$\begin{aligned} \lambda u_z(0,t) &= Q \cdot \delta(t) \\ u_z(l,t) &= 0 \end{aligned} \tag{2}$$

34

where λ is the thermal conductivity [W m⁻¹ K⁻¹], $\delta(t)$ is the Dirac delta function and Q the strength of the pulse [J]. The initial condition is:

$$u(z,0) = 0 \tag{3}$$

2

4 Solving through the application of the Laplace Transform, it is possible to obtain the 5 temperature evolution in time for each point of the slab, as described in Eq. (4):

$$u(z,t) = \frac{Q}{\lambda} \frac{\alpha}{l} \left[1 + 2 \sum_{n=1}^{\infty} (-1)^n e^{-n^2 \pi^2 \frac{\alpha}{l^2} t} \cos \frac{n\pi}{l} (z-l) \right]$$
(4)

5 6

that may be rewritten also in the form of Eq. (5):

$$u(z,t) = \frac{Q}{\epsilon\sqrt{\pi t}}e^{-\frac{z^2}{4\alpha t}} \left[1 + 2\sum_{n=1}^{\infty}e^{-\frac{(2nl)^2}{4\alpha t}}\cosh\frac{4nzl}{4\alpha t}\right]$$
(5)

7 8

In Eq. (5), ε is the thermal effusivity [J K⁻¹ m⁻² s^{-1/2}]. defined as in Eq. (6):

$$\epsilon = \sqrt{\lambda \rho c_p}$$
(6)

9

where ρ is the density [kg m⁻³] and c_p is the specific heat [J kg⁻¹ K⁻¹] of the material. It is 14 worth noting that Eq. (4) represents the deviations from the solution at the end of the process, 15 that is the behavior of u(z,t) when $t \rightarrow \infty$, while Eq. (5) represents the deviations from the 16 17 behavior of the semi-infinite body, that is u(z,t) when $t \rightarrow 0$. Defining the Fourier number Fo 18 as in Eq. (7):

$$Fo = \frac{\alpha t}{l^2} \tag{7}$$

15

it is possible to represent the results of Eq. (4), setting the value of the leftmost parameter 19 20 group $(Q\alpha/\lambda l)$ equal to 1. This is done in Fig. 2, where the x-axis is the normalized thickness 21 (z/l) of the slab and the y-axis is the ratio between the temperature T and the T_{inf} temperature 22 for $t \rightarrow \infty$.



the parameter group $(Q\alpha/\lambda l)$ is equal to 1, the x-axis is the normalized thickness (z/l) of the slab. A significant temperature increase may occur on a part of the specimen and may lead to undesired effects.

28 The diagram shows that a significant temperature increase may occur on a part of the 29 specimen and may lead to undesired effects. For example, materials such as Zirconium oxides 30 or Yttrium oxide partially stabilized Zirconium (YPSZ), that have low thermal effusivity 31 values with respect to metals, show a temperature increase on their front faces up to four 1 times higher than a metal (e.g. an AISI304 austenitic steel) [15]. Furthermore, the heat pulse 2 may create a high temperature gradient inside the specimen due to the low thermal diffusivity 3 of YPSZ. The combined effect of high temperature increase and temperature gradient within 4 the sample may lead, for certain values of the toughness of the specimen, to microstructural 5 modifications of the sample itself such as growing or reopening of microcracks [15]. A high 6 temperature gradient may also introduce some errors in the measurement of thermal 7 properties with a strong temperature dependence. A high temperature increase leads also to a 8 non-negligible heat exchange with the environment.

9 The non-adiabatic model of the Laser Flash method, sketched in Fig. 3, is always based on 10 Eq. (1) where boundary conditions of Eq. (8) are chosen, and a pulse of finite duration t_h is 11 considered:



of the temperature acquisition, and the pulse duration t_h that is set by the heating source (laser

equipment). 31 2.2 Data analysis

29

1 To gain a better understanding of the data analysis procedure we introduce an abstract model. 2 Let us consider *n* observations (y_i, x_i) , i=1,...,n, where the dependent variables (measurements) $v_i \in \mathbb{R}$ are related to the independent variables (regressors) $x_i \in \mathbb{R}^m$ via a 3 4 (generally) nonlinear relationship $y_i = g(x_i, \theta^*) + \varepsilon_i$, where ε_i is an unobserved noise. The 5 functional relationship g(t) is known modulo the components of the vector $\theta^* \in \mathbb{R}^p$, which are 6 fixed but unknown parameters, to be estimated on the basis of the observations. The 7 correspondences between the abstract formulation and Eq. (9) are as follows, $x_i = t_i$ is the *i*-th 8 sampling time, $y_i = T(t_i) + \varepsilon_i$, and $\theta^* \in \mathbb{R}^3$ with:

$$(\theta_1^*, \theta_2^*, \theta_3^*)^T = \left(\frac{Q}{h}, \frac{\alpha}{l^2}, B\right)^T$$
(12)

Define $\mathbf{y} = (y_1, ..., y_n)^T$, $g_i(\theta) = g_i(x_i, \theta)$, $\mathbf{g}(\theta) = (g_1(\theta), ..., g_n(\theta))^T$, where $\theta^* \in \mathbb{R}^p$, and $S(\theta) = (g_1(\theta), ..., g_n(\theta))^T$. 9 $||\mathbf{y} \cdot \mathbf{g}(\theta)||^2 = \sum_{i=1}^n (y_i - g_i(\theta))^2$, the function of θ measuring the goodness of fit of model $\mathbf{g}(\theta)$ 10 to the measurements y. The least squares estimator $\hat{\theta}$ of the parameter θ^* is defined as the 11 minimizer $\hat{\theta} \coloneqq argmin_{\theta}S(\theta)$. Imposing the first order conditions for the minimum one gets 12 13 the Eq. (13):

$$\frac{\partial}{\partial \theta_j} S(\theta) = 2 \sum_{i=1}^n (y_i - g_i(\theta)) \frac{\partial}{\partial \theta_j} g_i(\theta) = 0$$
(13)

14 15

which, letting $G = ||\frac{\partial}{\partial \theta_j} g_i(\theta)||_{ij} \in \mathbb{R}^{p \times p}$, can be written in matrix form as (normal equations) Eq. (14):

16

$$G^{T}(\boldsymbol{y} - \boldsymbol{g}(\boldsymbol{\theta})) = 0$$
⁽¹⁴⁾

17

These equations in θ do not have a closed form solution and iterative schemes need to be 18 setup to find $\hat{\theta}$. On the other hand the solution is straightforward in the linear case, i.e. when 19 $g(\theta) = X\theta$ for some $X \in \mathbb{R}^{pxn}$, and it is worth deriving it as it sheds light on the general case. In 20 21 the linear case one trivially gets G = X, thus the normal equations are $X^{T}(y - X\theta) = 0$ and their solution is $\hat{\theta} = (X^T X)^{-1} X^T y$. In case of repeated observations y_b t=1,...,N, where each 22 $y_t \in \mathbb{R}^n$, there are two equivalent ways of computing the estimator. In the first way one starts 23 computing the N estimators, $\hat{\theta} = (X^T X)^{-1} X^T y_t$, for t=1,...,N, and then computes their 24 average $\hat{\theta}_A \coloneqq \frac{1}{N} \sum_{t=1}^{N} \hat{\theta}$. Alternatively, one starts computing the averaged observations 25 $\overline{\mathbf{y}} \coloneqq \frac{1}{N} \sum_{t=1}^{N} \mathbf{y}_t$ and then computes the estimator $\hat{\theta}_B = (X^T X)^{-1} X^T \overline{\mathbf{y}}$. By inspection of the 26 27 above formulas it is immediate to verify that $\hat{\theta}_A = \hat{\theta}_B$. The advantage of the N repeated observations is to reduce the variance of the estimator by a factor 1/N. In the nonlinear case 28 29 the two procedures for dealing with repeated observations are not equivalent. On the other hand when (as is usually the case) the estimators are consistent, i.e. $\hat{\theta} \rightarrow \hat{\theta}^*$ for $n \rightarrow \infty$, since in a 30 31 neighborhood of θ^* the nonlinear model is well approximated with its linearized version 32 $g(\theta) \approx g(\theta^*) + G(\theta - \theta^*)$, the equivalence of the two procedures for repeated observations 33 still holds.

34 A good reference for this section is Seber [24], to which the interested reader is referred 35 for the missing mathematical details.

36 2.3 Experimental layout

37 The laser flash apparatus allows the execution of different types of experiments. In the basic 38 configuration the bell iar is lifted and the experiment is performed at room temperature and 39 atmospheric pressure. Including in the setup the bell jar, it is possible to perform experiments 40 in a controlled environment (flowing gas such as Nitrogen or Argon) or in vacuum, up to 10^{-6} 41 Torr. Another option is including a furnace, that could drive the specimen up to 1600 °C. Fig. 42 4 shows the experimental layout including the main elements.



Fig. 4. The experimental layout consists (from right to left) in a laser, a bell jar containing the furnace and the sample holder, and an infrared detector. The data acquisition system is placed on the back side of the apparatus.

18 The experimental layout includes six elements: a laser, a sample holder, a bell jar, a 19 furnace, a detector, and a control and data acquisition system. The laser is a Nd doped YAG 20 solid state laser (wavelength 1064 nm) pumped by two xenon filled flash lamps. The power supply is capable of rep-rates of over 600 pulses per minute with the output energy 21 22 continuously adjustable from the lasing threshold to at least 30 joules by varying the flash 23 lamp voltage and the lamp pulse width. The lamp voltage is adjustable from 0 to 1000 volts. 24 The pulse width is adjustable from 10 microseconds to 2 milliseconds. The sample holder is a 25 graphite and molybdenum ring, and is placed inside a bell jar. The bell jar has two infrared 26 transparent windows and is connected to a dual stage vacuum system and to a gas flowing 27 circuitry. The bell jar could also host a furnace, that consists in a tantalum foil and is monitored by a Platinum-Rhodium Thermocouple. The detector is a Teledyne J10D (InSb, 28 range 2-5.5 µm) operating in photovoltaic mode and connected to a P9 transimpedance 29 30 amplifier.

Before the experiments, the samples may be painted with a high emissivity black varnish
(Kontakt Chemie Graphit 33) to improve the thermal detector measurement and the
absorption of the laser pulse on the opposite side respectively.

22 3. The novel experimental setup

The existing Laser Flash setup has been modified introducing a new hardware and software
setup. In particular, the data acquisition system and the experiment control system have been
changed. The underlying idea is making possible the repetition of multiple measurements on
the same sample.

36 The architecture of the system is shown in Fig. 5a and includes: a computer with 37 dedicated software, an analog input/digital output controller, a laser controller, a laser, an 38 infrared sensor. In Fig. 5a the base architecture is represented on the right column, where the 39 laser controller activates the laser heating the specimen. The temperature variation is detected 40 by an Infrared sensor. The improved architecture is based on an analog input/digital output 41 controller, realized with a National Instrument NI 6211 module. This device has a 16-bit 42 analog input (sampling up to 250 kS/s) with an adjustable input range up to $\pm 10V$. The device 43 has an internal base clock up to 80 MHz and a digital output (TTL 5V). The device has been 44 connected on its analog input channel to the infrared detector (described in the next section) 45 and on its digital output channel to the laser controller.



showing the measurement variables available for the user.

16 A new software developed in Labview has been created for the measuring process. The 17 user interface is shown in Fig. 5b. The software relies on the Dagmx SubVIs, that manages 18 the input and output channels of the data acquisition board. The user can choose the input 19 range and the sampling rate, that is adjusted on the estimated signal coming from the detector. 20 The buffer size is chosen to maximize the timing accuracy of the board. The user can also 21 choose when the laser shot is performed and the duration of the shot. Another important 22 feature for the user is the possibility of repeating the measurement, choosing also the delay 23 between repetitions. At the end of the repetitions, the user obtains the data of all the 24 repetitions and the average value.

18 The number of repetitions and the delay between each repetition should be chosen taking 19 into account both the kind of material under test and the experimental conditions (specimen temperature, environment). Following the model described in section 2, the user could predict the specimen behavior and assess the measurement uncertainty related to the temperature of the specimen [21].A first test of the hardware and software has been made short-circuiting the input and the output of the board. The second testing round has been a long set of measurements on a sample with the possibility of detecting also the laser signal. This has allowed the quantification of a repeatable delay between the output from the control board and the actual laser shot.

8 4. Materials and methods

9 The two types of tested specimens are part of an ongoing research on thermal barrier 10 coatings: the first is made of AISI 304 stainless steel, that is chosen as a reference for the 11 apparatus testing, while the second is made of Zirconium oxide. The Zirconium oxide (ZrO₂) 12 specimen has been thermally treated in an oven at 1100 °C for 300 hours to simulate the 13 aging of the material under working conditions. The series of measurements are listed in Tab. 14 1. The number of repetitions is chosen to increase of a desired quantity the SNR.

15 16

Series	Sample	Environment	Number of repetitions
#1	AISI 304	Room temperature	100
#2	ZrO ₂	Room temperature	16
#3	ZrO ₂	Room temperature, vacuum	16
#4	ZrO ₂	High temperature (1100°C)	16

17 The AISI 304 sample has a thickness of 1.024 mm, while the Zirconium oxide sample 18 thickness is under measurement in the project. Therefore, in this paper the results for AISI 19 304 are listed as thermal diffusivity, while the results for Zirconium oxide are expressed as 20 the output of the fitting parameter α/l^2 .

21 For each series, two averaging methods are performed and compared. Firstly, the data of the average have been fitted with a nonlinear solver (implementing the Levenberg-Marquardt 22 23 algorithm) available in the Matlab environment. Then the obtained parameters have been used 24 as the starting point for the fitting of each repeated measurement. The saving in 25 computational time is remarkable when performing the fit of the averaged value with respect 26 to fitting each measure and successively averaging the parameters. This is due to two reasons that have a different effect on the calculation. The first is that the fitting process is the most 27 time-consuming calculation, therefore removing the need of repeating it decreases the time 28 proportionally to the number of repetitions. The second reason is that the speed of the fit 29 30 typically decreases, going from up to a minute to just few seconds on the same computing 31 machine (MacBook Pro with 2.6 GHz dual-core intel i5 and 8 GB RAM), when the data have 32 a low SNR. Therefore, the fit of any single measurements (one shot) is more time consuming 33 than the fit of the average shot.

For each fit, the root mean square of the differences between the experimental data and the fit is chosen as an indicator of the measurement noise [16]. Then the SNR is defined as the ratio between the maximum value of the fitted signal and the aforementioned noise.

37 5. Results

The results for the AISI 304 specimen are shown in Tab. 2, where it is possible to note that the two averaging processes described above lead to the same value of thermal diffusivity, a result that is compatible with the literature references [17].

Table 2. Results for the measurement series #1 on AISI 304. *single fit

Series and Method	Estimated parameter α/l^2 [s ⁻¹]	Thermal diffusivity [m ² s ⁻¹]	Standard deviation of the fitting error [a.u.]	Signal- to-noise ratio [a.u.]
#1 AISI 304	3.96	$4.1 \cdot 10^{-6}$	$5.4 \cdot 10^{-5}$	24.2
Fit on the average value				
#1 AISI 304	3.96	$4.1 \cdot 10^{-6}$	$4.8 \cdot 10^{-4}$	2.7*
Average of the fits on each repetition				

The noise level on a single measurement is, as expected, higher than the averaged value, theoretically by a factor equal to the square root of the number of measurements. A better comparison of the noise level is proposed in Fig. 6, where the fit on the average value (average shot) and the fit on a single measurement (single shot) are plotted against the experimental data.



Fig. 6. Results of the AISI 304 specimen measurements. Experimental data and fitting curve for a single shot (a) and for the average of 100 shots (b). The noise level is significantly lower for the average (b).

The lower noise of the averaged value improves the quality and the calculation speed of the fit. The results for Zirconium oxide are shown in Tab. 3, for the different experimental conditions.

Table 3. Results for the measurement series #2-#4 on ZrO ₂ . *single	fit

Series and Method	Estimated parameter α/l^2 [s ⁻¹]	Standard deviation of the fitting error [a.u.]	Signal-to-noise ratio [a.u.]
#2 ZrO ₂ - Room temperature	0.30	$1.9 \cdot 10^{-4}$	71.3
Fit on the average value			
#2 ZrO ₂ - Room temperature	0.30	$5 \cdot 10^{-4}$ *	27.1*
Average of the fits on each repetition			
#3 ZrO ₂ - Room temperature - Vacuum	0.29	$1.2 \cdot 10^{-4}$	38.1
Fit on the average value			
#3 ZrO ₂ - Room temperature -	0.29	$4.8 \cdot 10^{-4*}$	9.8*

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Vacuum			
Average of the fits on each repetition			
#4 ZrO ₂ - 1100 °C	0.19	$2.2 \cdot 10^{-4}$	136.6
Fit on the average value			
#4 ZrO ₂ - 1100 °C	0.19	$6.2 \cdot 10^{-4*}$	48.3*
Average of the fits on each repetition			

As expected, also in this case the estimation of the parameter is equal for the fit on the 2 average value and for the average of the fits on each repetition. The decrease of the signal-tonoise ratio for the room temperature measurement in the vacuum case is due to the presence of the bell jar, as the infrared windows before and after the sample contribute to a decrease of the signal. For the ZrO_2 the decrease of the estimated parameter for the high temperature experiment is expected, see also [18]. Figure 7 shows, as for the AISI specimen, the improvement of the averaged measurement compared to a single one. With respect to Fig. 6, 7 8 the signal is stronger due to the high temperature measurement.





13 The experiments presented in this work have a sufficient SNR to enable the process on a 14 single fit and have been chosen to demonstrate the equivalence of the obtained result. 15 Working with lower laser inputs or with specimens having more challenging configurations (e.g. low thermal diffusivity or investigating thick samples) may lead to the impossibility of 16 performing a fit on a single profile, leaving only the fit on the average profile as a feasible 17 18 method

19 Conclusions 6.

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20 The existing laser flash setup has been improved with the creation of a software in Labview 21 environment and the use of a new data acquisition and control board. It is now possible to 22 repeat automatically a preselected number of measurements on a single specimen, improving 23 significantly the signal-to-noise ratio of the measure in an easily manageable way. Results on different materials (AISI 304 and ZrO₂) and different testing conditions (room temperature, 24 vacuum, high temperature) showed the predicted increase in SNR due to the proposed 25 averaging method. This setup facilitates working with low laser input power and with 26 27 specimens that have an unfavorable combination of thermal properties and thickness. Future work will investigate the mathematical modeling, to determine the optimal data treatment for 28 29 the averaging of multiple measurements.

30 **Disclosures**

31 The authors declare no conflicts of interest.

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