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1 **Fast-tracking of NH₃ interaction with ZnO nanorods and C/ZnO hybrid nanostructures by**
2 **operando spectroscopy**

3
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17
18 **Abstract**

19 Fast X-ray photoemission spectroscopy has been carried out to track the interaction of ammonia gas
20 with hybrid materials based on ZnO nanorods and nanostructured carbon. We provide, for the first
21 time, direct evidence of gas adsorption of ammonia on nanostructured hybrid interfaces combining
22 photoemission with transport measurements through the resistivity change upon gas exposure. This
23 operando methodology allowed us to reconstruct the adsorption isotherms and relate the different

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1 behavior to the sample morphology at the nanoscale. While on bare ZnO nanorods, physisorption of
2 ammonia is detected along with photon beam assisted cracking, on the nanostructured carbon the
3 formation of amine groups is detected, without displaying any effect induced by photon beam
4 exposure. Our results make the present experiment an operando general methodology, with the
5 possibility to relate the transport versus dose curves (ultimately the adsorption isotherms) to a specific
6 investigation of the gas/surface interactions.

7

8 **Keywords:** ZnO, C/ZnO hybrid, NH₃ gas, fast acquisition XPS, adsorption isotherms, resistance

9

10 **1. Introduction**

11 Gas sensing represents one of the most important fields where carbon/metal oxide hybrids
12 nanostructures are efficiently applied [1]. Carbon nanostructures (CNs) and nanostructured metal
13 oxides, such as nanorods (NRs), are both individually used to detect gases and they are excellent
14 complementary materials to create heterojunctions [2-8]. When combined in hybrid structures, they
15 compensate for the shortcomings of the single components and their combination creates new
16 advantageous features. Despite intensive research to advance the understanding of these materials,
17 the complex array of possible CNs shapes and different metal oxides present several virtually
18 unexplored areas. Specific real-time spectroscopic study of adsorption processes aimed to identify
19 the dominant interactions between gas molecules and nanostructured surfaces and hybrids are still
20 rare. Operando methodologies would enable the investigation of the adsorption isotherms, relating
21 them to the sample morphology at the nanoscale.

22 Among carbon/metal oxide hybrids, heterointerfaces of ZnO and CNs have received much attention
23 [9-14] due to the possibility to tune the morphology of the samples through the binary ZnO-C system
24 down to the nanoscale. Furthermore, the capability to adsorb molecules on the surface makes these
25 hybrids also interesting for gas storage applications and, in our particular case, for ammonia storage
26 [15].

1 In this paper, the adsorption and desorption chemistry of ammonia gas on the ZnO NRs and CNs/ZnO
2 hybrid grown using ZnO NRs as a template (C/ZnO) were studied by fast-acquisition XPS. The high
3 energy-resolution of the photoemission probe used, along with the high photon flux and the tunable
4 photon energy, allowed us to understand some of the mechanisms governing the interaction of the
5 C/ZnO hybrid nanomaterials with NH₃ gas. Our data showed that ammonia chemisorbs on the hybrid
6 C/ZnO structure, while it physisorbs on the bare ZnO NRs.

7 While acquiring fast-XPS data, the samples electrical resistance was collected and the adsorption
8 isotherms were determined by considering the resistance change of the hybrid layers upon ammonia
9 adsorption. This makes the present experiment an operando methodology, with the possibility to
10 relate the transport versus dose curves (ultimately the adsorption isotherms) to a specific investigation
11 of the NH₃/surface interaction. The adsorption isotherms showed for the bare ZnO NRs a steady
12 increase and reached saturation at 400 L (1 L = 10⁻⁶ Torr), while for the C/ZnO hybrid the isotherms
13 displayed two steps in the 0-400 L range. The curve of the C/ZnO hybrid displayed features quite
14 similar to those observed for NH₃ chemisorption on mesoporous carbon [16], while, the curve of the
15 bare ZnO nanorods displayed a behavior similar to that of physisorption by microporous solids [17].

16

17 **2. Experimental**

18 ***2.1 Synthesis of ZnO and C/ZnO hybrid nanostructures***

19 The ZnO NRs were hydrothermally synthesized using a method developed by Wang et al [18]. This
20 method involves the deposition of a thin film of ZnO on a silicon substrate prior to the hydrothermal
21 growth. The film acts as the nucleation site for the NRs growth and ensures the vertical alignment of
22 the grown NRs, as described in details in [19]. To synthesize the C/ZnO hybrid, the CVD process
23 was carried out on the ZnO NRs at 800 °C using 50 sccm C₂H₂, see [20] for more details.

24

25 ***2.2 Characterization***

1 The ZnO NRs and C/ZnO hybrid were characterized by SEM and XPS, ex-situ. The morphology of
2 the nanostructures was studied using a 30 μm aperture SEM at 10 kV. The interaction of NH_3 gas
3 with the nanostructures, was investigated by fast acquisition mode XPS during gas exposure and
4 desorption using synchrotron radiation at the beamline for advanced circular dichroism (BACH) at
5 Elettra. The XPS spectra were acquired at a rate of 300 ms per spectrum using a calibrated Scienta
6 R3000 electron analyzer [21]. The photon energy could be tuned between 45-1600 eV with high
7 resolving power of 2000-6000 in the entire energy range. Each sample was first heated in the
8 preparation chamber to remove surface contaminants ($\sim 350\text{-}400^\circ\text{C}$), and then moved to the
9 measurement chamber, where it was exposed to NH_3 at RT at different pressures (in the 10^{-7} - 10^{-5}
10 mbar range). We performed the experiment using a leak valve, hence the range in pressure, we tried
11 to expose the sample to the lowest possible pressure because for environmental monitoring it is
12 important to detect the lowest possible concentration of a gas. During gas exposure the electrical
13 resistance of the sample was measured, along with the C 1s, Zn 3d, O 1s, and N 1s XPS core level
14 spectra.

15 Photon damage was checked and excluded by systematically moving the beam spot in sample areas
16 not yet illuminated by the beam. After the measurements the sample was heated to remove the
17 adsorbed ammonia, checking the decrease of the N1s intensity until its complete disappearance.

18

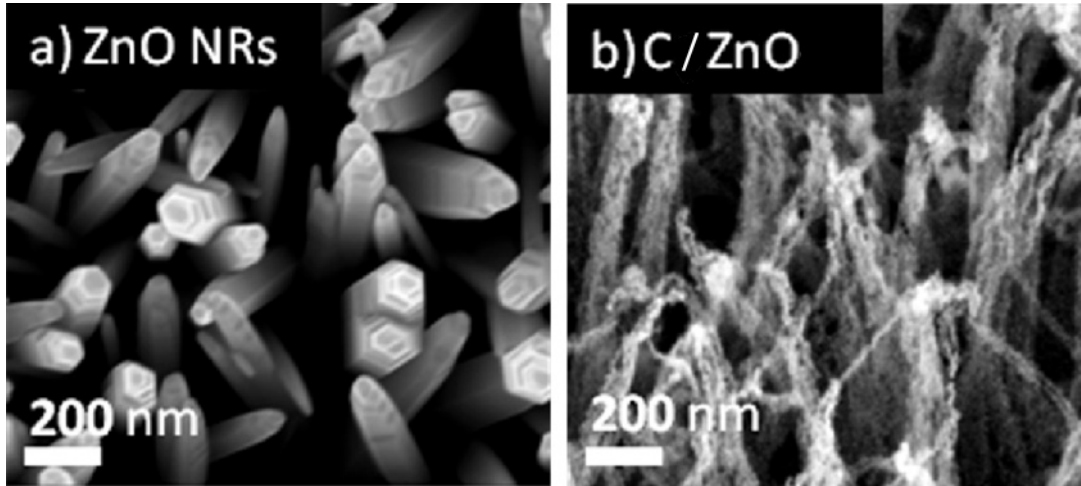
19 **3. Results and discussion**

20 ***3.1 Morphology***

21 The deposition of a thin ZnO film on the silicon substrate prior to the hydrothermal process resulted
22 in the growth of vertically aligned ZnO NRs with a pointed hexagonal shape at the end of the c-axis
23 as previously described [18, 20], and shown in Figure 1a. The shape of the NRs is due to the etching
24 effect during growth [18, 22]. **The as grown nanorods are $\sim 1\mu\text{m}$ in length and their diameter vary in**
25 **the range of 50 to 200 nm.** The C_2H_2 CVD process on the ZnO NRs resulted in the etching of the
26 ZnO NRs and in the synthesis of 1-D carbon nano-dendrites, preferentially aligned along the location

1 of the ZnO nanorods [18], see Figure 1b. During the CVD process C_2H_2 decomposes to carbon and
2 hydrogen and reacts with the ZnO nanorods forming adsorbed carbon and decomposition products
3 which includes Zn, H_2O , CO and CO_2 , which are in the gas phase during CVD

4



5

6 **Figure 1:** SEM micrographs of a) ZnO NRs grown by the hydrothermal method, b)
7 C/ZnO hybrid nanostructures grown at 800 °C using the ZnO NRs as template.

8

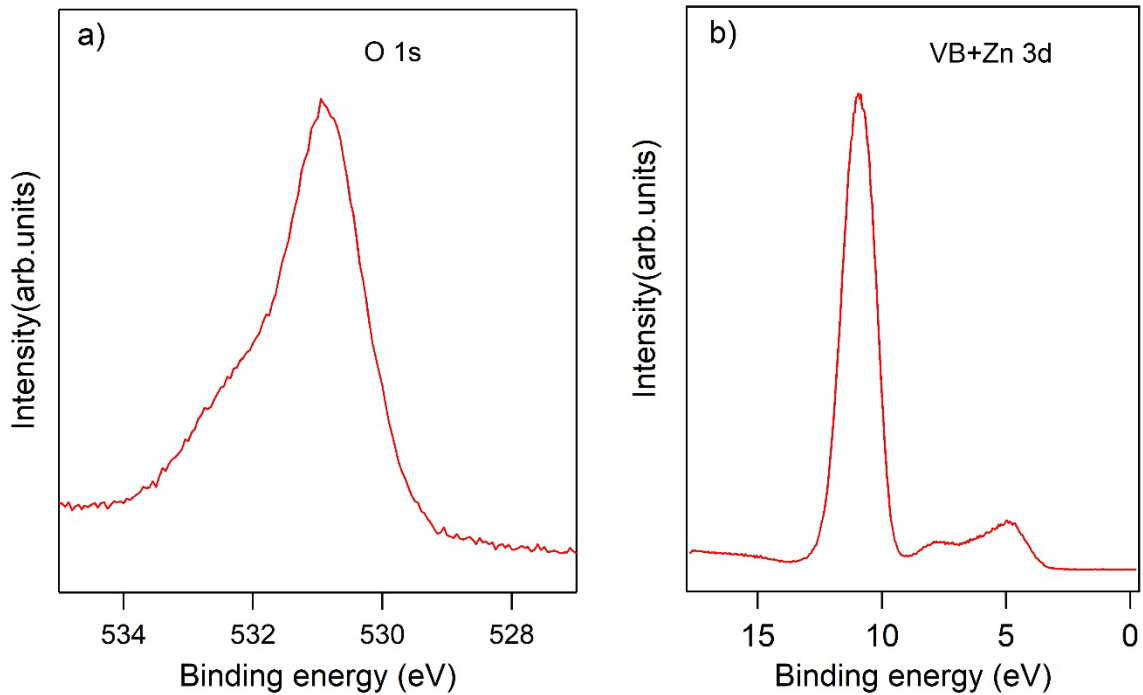
9 3.2 Fast XPS during ammonia gas exposure

10

11 3.2.1 Bare ZnO NRs

12 The ZnO NRs were first annealed at ~350-400 °C in UHV (Ultra High Vacuum) before ammonia
13 exposure to remove contaminants from the hydrothermal process and air exposure. The O 1s spectrum
14 of the ZnO NRs after annealing (Figure 2a) is mainly composed of two peaks at binding energy (BE)
15 530.85 and 531.95 eV which can be assigned to O of Zn-O and hydroxyl groups, respectively [23-
16 25].

17



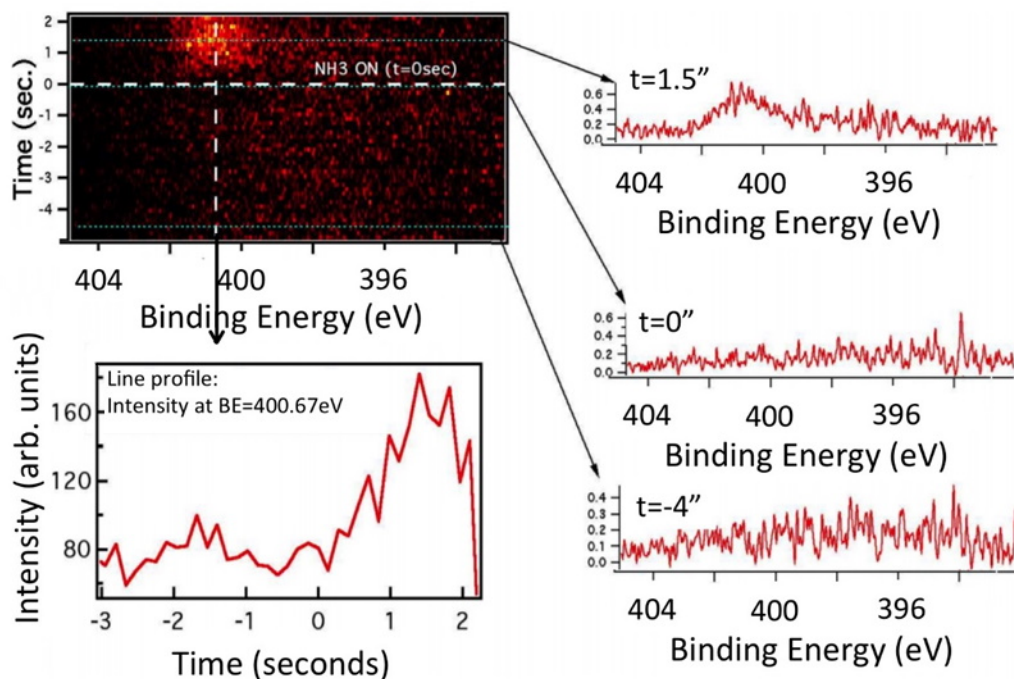
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2 **Figure 2:** XPS spectra of a) O 1s and b) valence band, including the Zn 3d level at ~11 eV, of
 3 the ZnO NRs after degassing at ~350°C. The spectra have been acquired using a photon
 4 energy of 605 eV, with an overall resolution of ~0.2 eV.

5

6 In Figure 3 (top left), a two-dimensional plot of the N 1s XPS spectra acquired using a photon energy
 7 of 605 eV during NH₃ gas exposure at ~10⁻⁵ mbar is shown, where the x-axis corresponds to the
 8 photoelectron BE, the y-axis to the exposure time (in seconds) and the figure color scale corresponds
 9 to the XPS intensity (black minimum, yellow maximum). On the right, single spectra extracted from
 10 the 2D plot (each acquired in ~300 ms) at selected significant time, before and during dosing, are
 11 shown. It is evident from Figure 3 that as soon as NH₃ starts to be dosed (time = 0 s) a peak at ~
 12 400.67 eV appears. The NH₃ dosing was stopped after ~2 s, in correspondence to the peak intensity
 13 dropping again to 0, as shown on the bottom left spectrum. According to literature [7], the peak
 14 observed at BE = 400.67 eV can be ascribed to physisorbed ammonia [26].

15

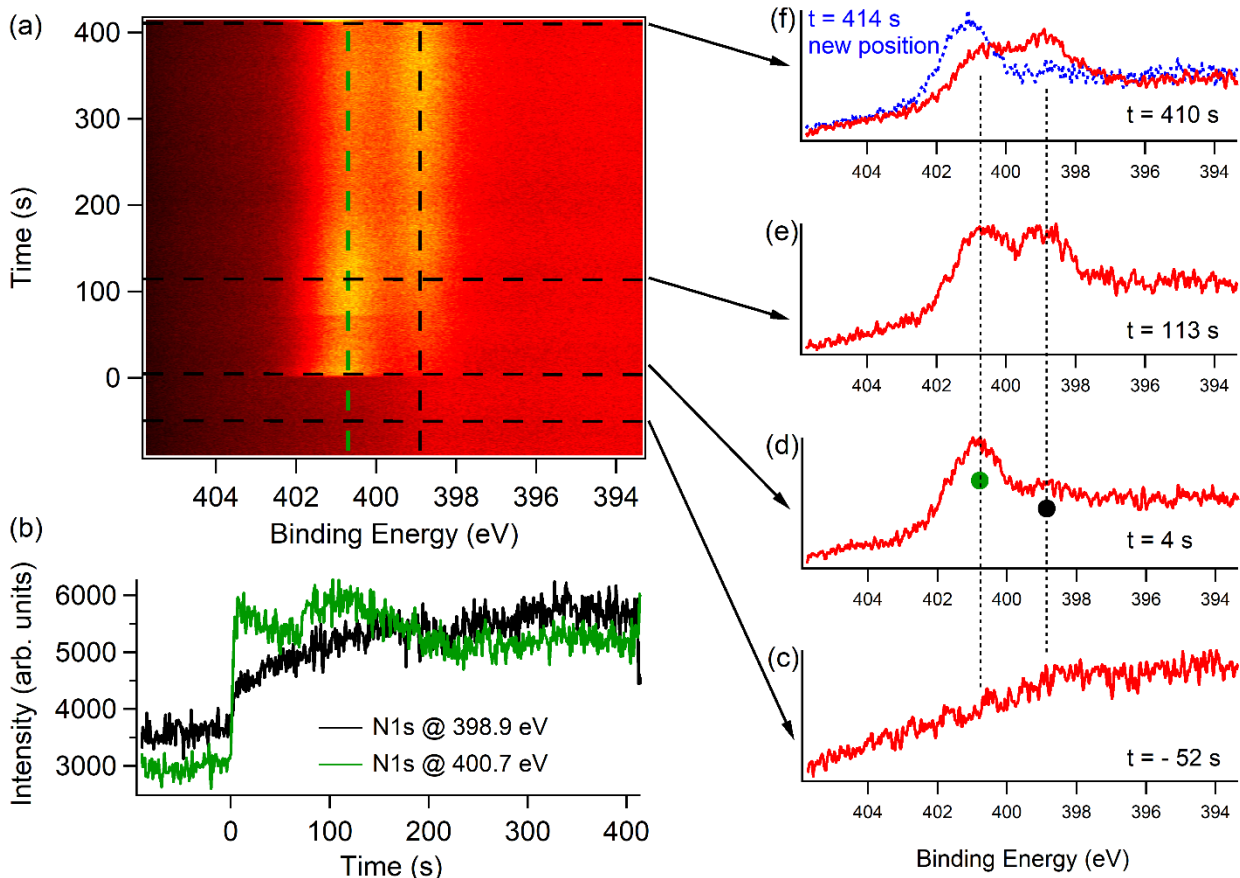


1

2 **Figure 3:** Two-dimensional fast XPS spectra of the N 1s level acquired at RT during NH₃
 3 dosing at $\sim 10^{-5}$ mbar (black: minimum intensity, yellow maximum intensity) on the ZnO NRs.
 4 Each spectrum has been acquired using an acquisition time of ~ 300 ms, with the analyzer in
 5 fixed energy mode at a pass energy of 50 eV. Time = 0 s corresponds to the time when gas
 6 exposure started.

7

8 When XPS measurements were performed during NH₃ dosing at lower pressure (5×10^{-7} mbar) and
 9 for longer time (~ 414 seconds), a progressive growth of an additional peak at lower BEs (~ 398.9 eV)
 10 was observed, as shown in Figure 4. The lower energy component's intensity increased until it
 11 becomes the dominant peak after 410 seconds of exposure.



1

2 **Figure 4:** (a) Two-dimensional fast XPS spectra of the N 1s level acquired at RT during NH₃
 3 dosing at $\sim 5 \times 10^{-7}$ mbar (black: minimum intensity, yellow maximum intensity) on the ZnO
 4 NRs. Each spectrum has been acquired in ~ 300 ms using the analyzer fixed-energy mode and
 5 a pass energy of 50 eV. (b) Vertical line profile showing XPS signal versus time related to peak
 6 energy of 400.7 eV (green line) and 398.9 eV (black line). (c-f) Horizontal line profile showing
 7 XPS spectra at different time (peaks at 400.7 eV and 398.9 eV are highlighted by dashed vertical
 8 lines and green and black dots in panel d).

9

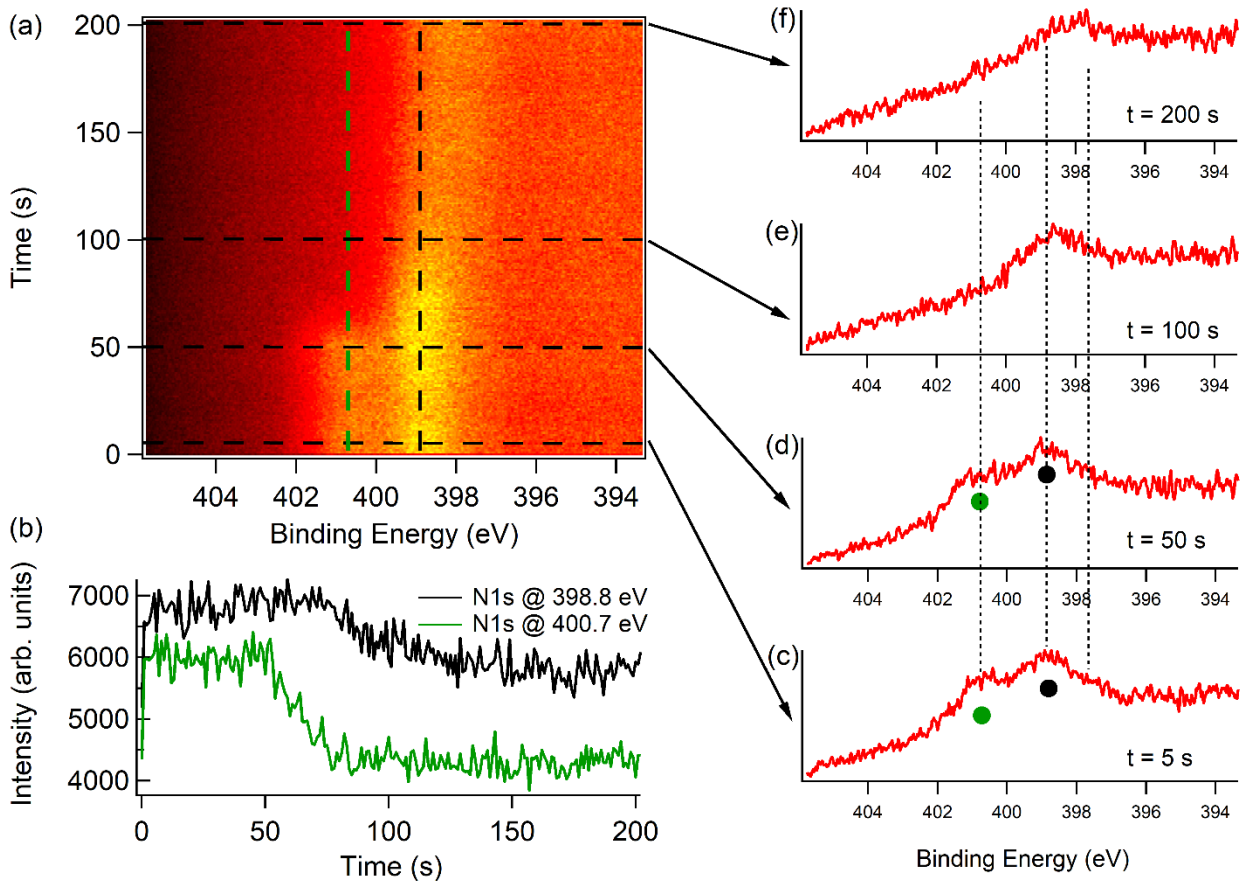
10 When the measuring position on the sample was changed, by displacing the sample holder by a few
 11 mm (see Figure 4, top right, blue spectrum), the higher energy component became again the dominant
 12 peak. This suggests that the photon beam has induced the cracking of the NH₃ molecules and the
 13 physisorption of these products on the ZnO NRs surface. The increase of the lower BE peak intensity
 14 with beam exposure duration is similar to that reported for electron bombardment [27] and heat [28]

1 stimulated dissociation of ammonia. We note that the N 1s peak observed after changing the sample
2 holder position (blue line Figure 4f) is shifted by ~ 0.5 eV to higher binding energy with respect to
3 the initial one ($t=4$ s, Figure 4d). A possible explanation can be that the sample was not homogeneous
4 on a scale compatible with the dimension of the photon beam ($\sim 250 \times 250 \mu\text{m}^2$). In addition the first
5 peak has been observed dosing during photon exposure, which can damage the adsorbed ammonia
6 on a time-scale lower than the spectra acquisition time (300 ms.).

7

8 In order to clarify this aspect and compare the thermal effect with that induced only by the photon
9 beam, fast XPS spectra of N1s core level were recorded on NH_3 -exposed ZnO during annealing at
10 100°C (see Figure 5). The analysis of the 2D fast XPS spectra of thermal treated sample in Figure 5
11 shows that the peak at 400.7 eV is less intense than that of unheated sample in Figure 4. This peak
12 completely disappeared after 100 s of heating at 100°C , while the intensity of the peak at 398.8 eV is
13 reduced but not completely quenched even after 200 s of heating. A new minor component at 397.8
14 eV, due to thermal decomposition of NH_3 , was also observed.

15



1

2 **Figure 5:** (a) Fast XPS spectra of the N 1s level with the corresponding intensity variations
 3 (black: minimum intensity, yellow maximum intensity) with time acquired during annealing
 4 the ZnO NRs sample at $\sim 100^\circ\text{C}$ (left panel). (b) Vertical line profile showing XPS signal versus
 5 time related to peak energy of 400.7 eV (green line) and 398.8 eV (black line). (c-f) Selected
 6 single spectra extracted from the 2D plot are also shown on the right panel. In addition to the
 7 peaks at 400.7 eV and 398.8 eV, also observed in the Figure 4, a minor component at around
 8 397.8 eV, due to decomposition products, is highlighted by a third dashed vertical line.

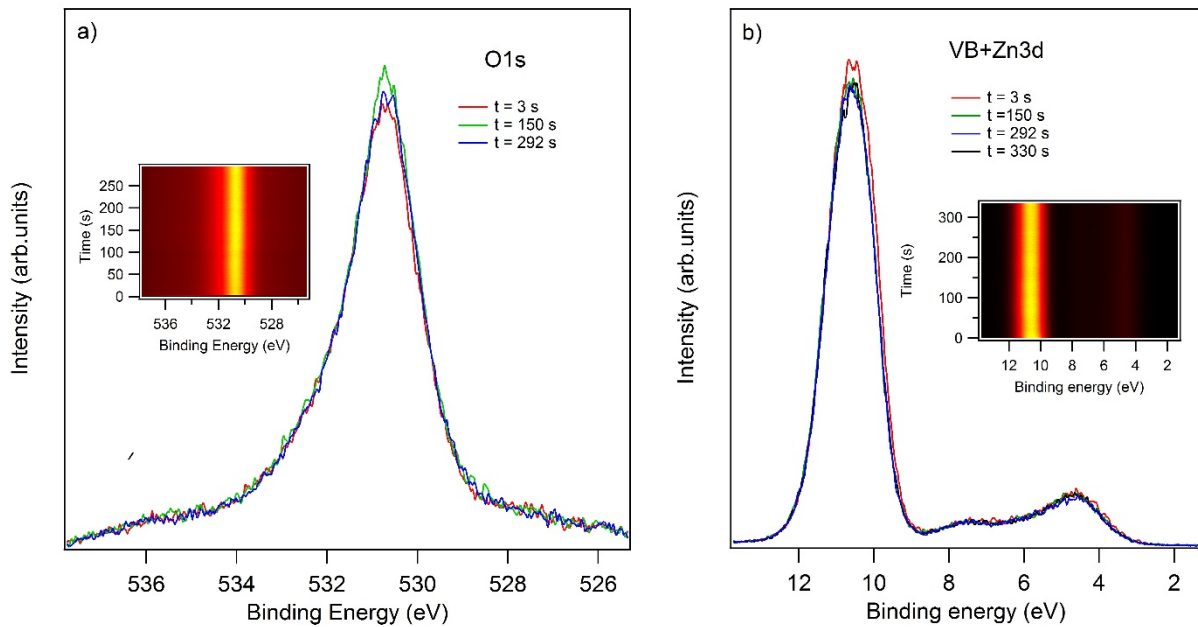
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10 Based on the results discussed above, we therefore assign the peak at 400.7 eV to physisorbed
 11 ammonia and the low BE N 1s peaks (the main at 398.8 eV and a minor component at around 397.8
 12 eV) to ammonia decomposition products [26, 29, 30].

13 In addition, as further confirmation of physisorption mechanism of NH_3 molecule on ZnO surface,
 14 the O 1s and Zn 3d peaks of ZnO were measured (Figure 6) during NH_3 exposure before the N 1s

1 saturation was reached. For both elements no change in the line shape or peak position was observed
2 or the formation of zinc nitride even after 330 seconds of exposure.

3



4

5 **Figure 6:** Fast XPS (a) O1s and (b) valance band plus Zn 3d level during ammonia exposure,
6 before the saturation of the N 1s signal was detected. Their corresponding 2D plots are reported
7 as inserts.

8

9 From these results we can conclude that ammonia is physisorbed on the ZnO NRs. The NH₃
10 physisorption process is very fast, detectable in a time-scale of ~100 milliseconds, and the sample
11 reacts in UHV at very low doses of NH₃, showing visible modifications in the N 1s core level upon
12 an exposure of 7.5×10^{-2} Langmuir (1 second at 10^{-7} mbar).

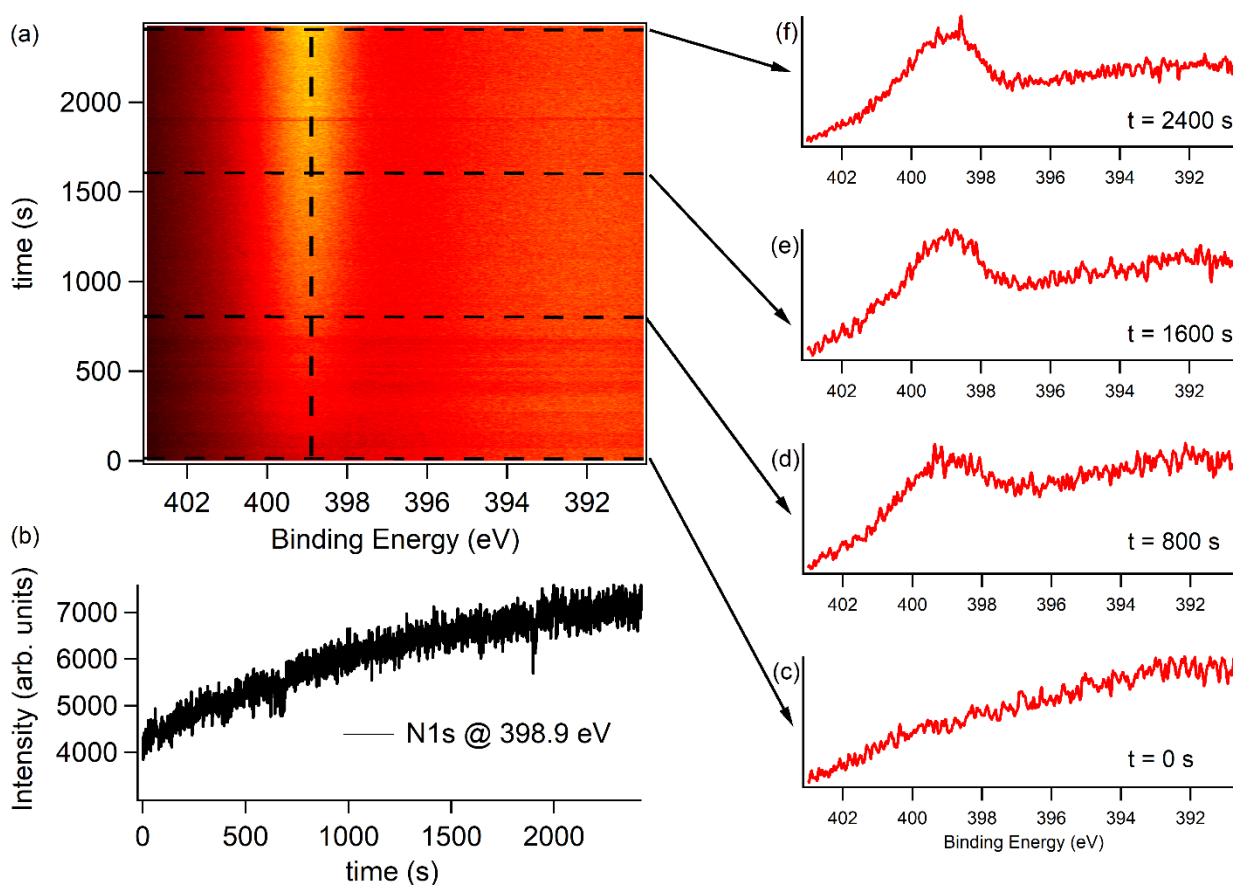
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14 3.2.2 C/ZnO hybrid nanostructures

15 When the C/ZnO hybrid was exposed to ammonia gas for ~ 2400 seconds at $\sim 10^{-7}$ mbar, the
16 chemisorption of ammonia with formation of amine groups was observed after about 800 sec. This is
17 indicated by the appearance of the N 1s peak at 398.9 eV, see Figure 7, which corresponds to

1 chemically bound nitrogen in the form of NH_x species [31], while no sign of NH_3 physisorption was
 2 detected. The chemical interaction between amine group species and C/ZnO hybrid is also visible in
 3 C1s core level (see Figure 8), where the broadening of the spectrum can be ascribed to the emergence
 4 of a new component at ~ 286 eV, most likely related to the formation of C-N bond [32]. The N 1s
 5 peak remained at the same BE even if the sample position was changed or if NH_3 was dosed without
 6 exposing the sample to the photon beam, using the same procedure as for the bare NRs. This indicates
 7 that the ammonia is not affected or dissociated by the photon beam, unlike the case of the bare ZnO
 8 NRs previously discussed. It should be noted that for this sample a lower resolution was used as
 9 compared to the NRs, hence the spectra appear broader than the previous ones.

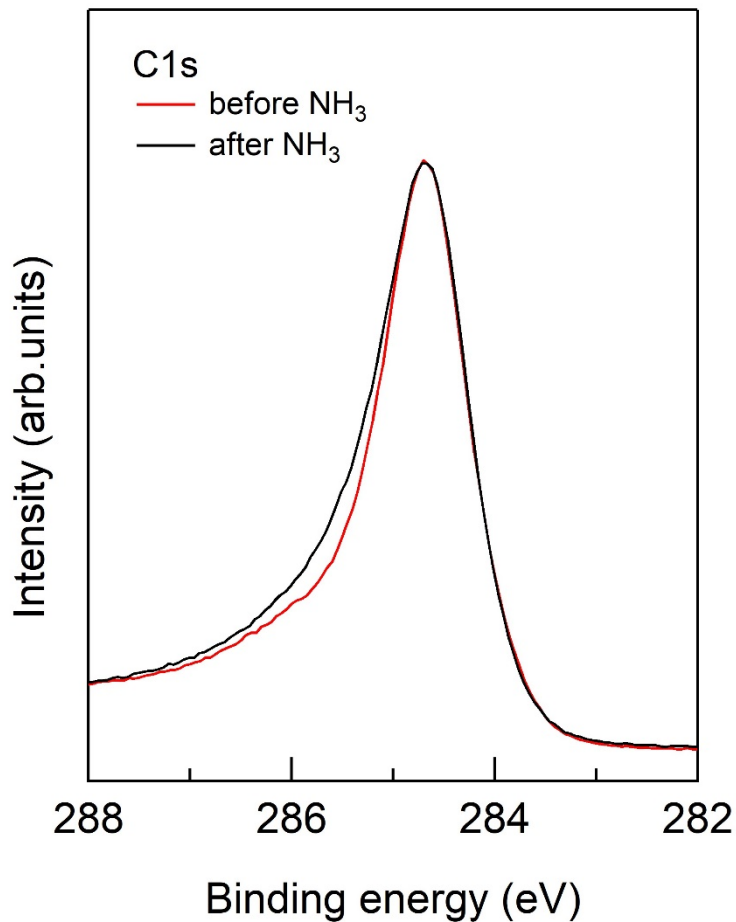
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11

12 **Figure 7:** (a) Two dimensional fast XPS spectra of the N 1s level acquired at RT during NH_3 dosing
 13 at $\sim 10^{-7}$ mbar of the C/ZnO hybrid grown at ~ 800 °C (black: minimum intensity, yellow maximum
 14 intensity). Each spectrum has been acquired in ~ 300 ms, using the analyzer fixed-energy mode, and

1 a PE=100 eV. (b) Vertical line profile showing XPS signal versus time related to peak energy of
2 398.9 eV. (c-f) Selected single spectra at different time during gas exposure.



3
4 **Figure 8:** XPS spectra of the C 1s level of the C/ZnO hybrid, before (red) and during
5 (black) $\sim 10^{-7}$ mbar ammonia exposure.

6
7 It was also noted that, when the C/ZnO hybrid was exposed to ammonia, the intensity of the
8 chemisorbed NH_x species in N 1s spectrum gradually increased, unlike the case of bare NRs, where
9 a fast conversion of physisorbed ammonia into cracked species, such as $-\text{NH}_x$, was observed. The
10 sample did not reach saturation even after ~ 2400 seconds of ammonia gas exposure. This may be
11 attributed to the difference in morphology of the two samples. Figure 1 shows that the hybrid is highly
12 porous and requires a longer time to reach saturation, unlike the solid ZnO NRs. In addition, XPS is
13 a surface sensitive technique: it only measures the N 1s uptake on the surface and not in the bulk. The

1 apparent slower response does not imply a slower and/or weaker response of these materials in their
2 bulk electronic properties, as will be shown by resistivity measurements in the next section.

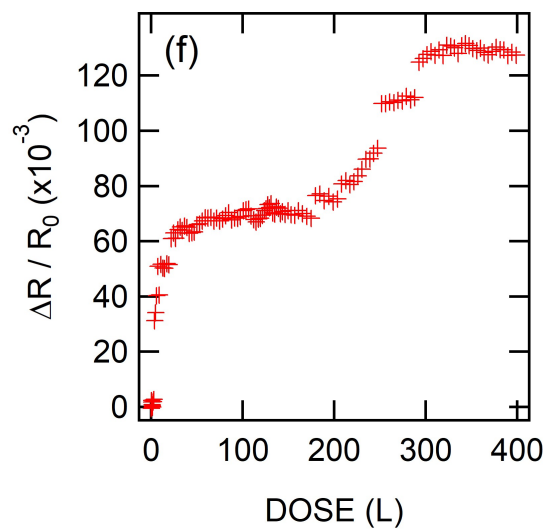
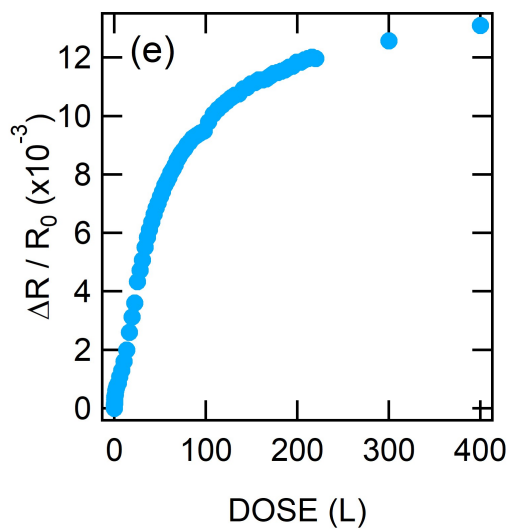
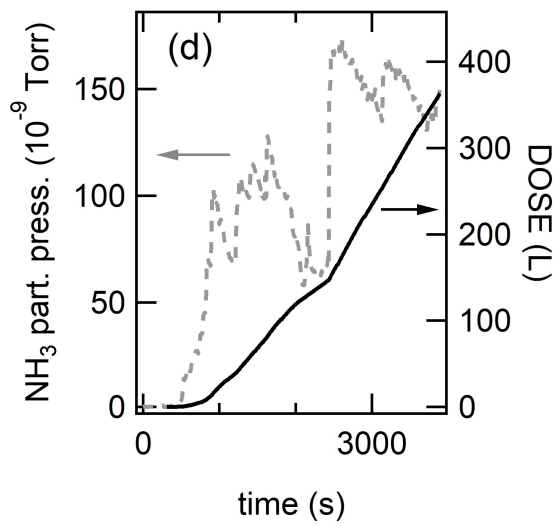
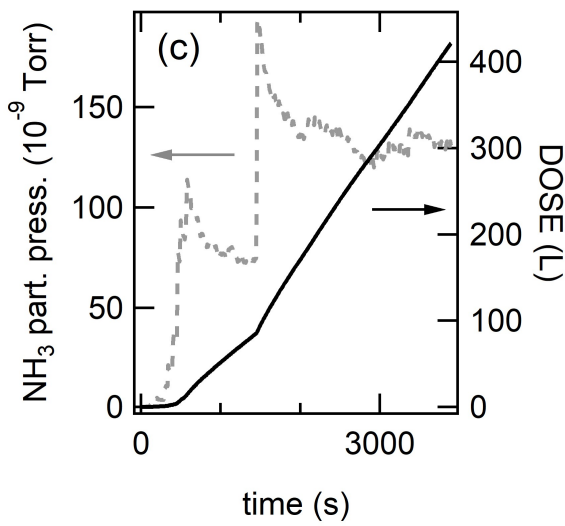
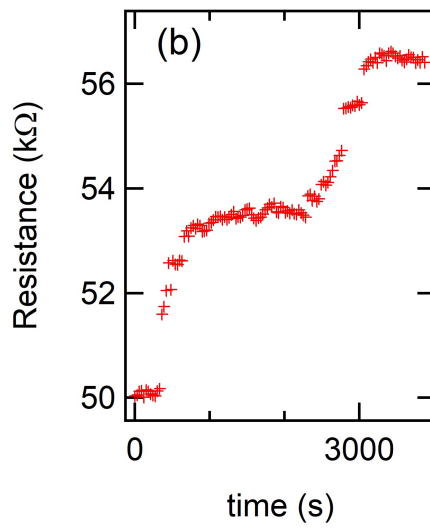
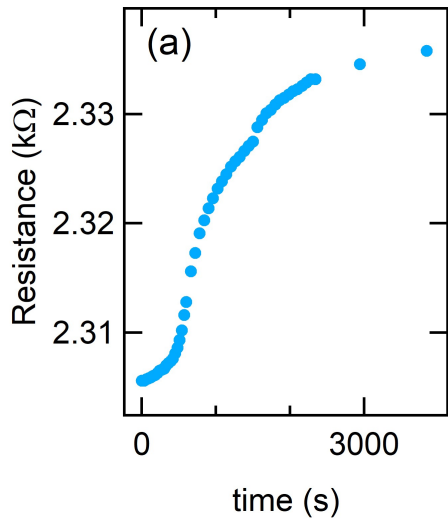
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4 **3.3 Adsorption isotherms**

5 During ammonia gas exposure, the sample electrical resistance variation was recorded. Figure 9
6 shows the resistance versus time during NH₃ gas exposure (a, b); ammonia partial pressure (grey
7 dashed line) and dose (black solid line) versus time (c, d); resistance relative response versus dose (e,
8 f) are shown, for bare ZnO NRs and C/ZnO sample, respectively. It should be noted that the response
9 $\Delta R/R_0$ of the C/ZnO hybrid sample is in general higher than for the bare ZnO NRs. In particular, at
10 400 L the response $\Delta R/R_0$ of the C/ZnO hybrid is nearly one order of magnitude larger than for the
11 bare ZnO NRs.

12 To analyze these results, we plotted the sample resistance variation versus the dose (L) of the gas
13 calculated by the integral of the partial pressure of gas versus time. If we assume that the response
14 $\Delta R/R_0$ is proportional to the amount of molecules adsorbed on the surface, and since the partial
15 pressure is proportional to the dose of the gas injected in the chamber, we can relate the curves
16 displayed in Figure 9 to the NH₃ adsorption isotherms on the two different surfaces. Under this
17 assumption, we observe that the measured isotherms (Figure 9e-f) are quite different in the two cases.
18 For the ZnO NR (Figure 9e) the isotherm steadily increases and reaches a saturation at 400 L, while
19 for the C/ZnO hybrid (Figure 9f) the isotherm displays two steps in the 0-400 L range. These
20 differences are usually ascribed to the different surface morphologies and are used for the
21 classification of gas-solid adsorption isotherms. Isotherms are classified according to the IUPAC
22 scheme into six different plots, depending on the mechanisms underlying the adsorption process and
23 the morphology of the solid [33]. In the present case the $\Delta R/R_0$ vs dose curves resembles those
24 classified as type I (ZnO NR, Figure 9e) and type VI (C/ZnO hybrid, Figure 9f). Though a detailed
25 study of the adsorption isotherms goes beyond the scope of the present study, we wish to observe that
26 the curve of Figure 9f displays features quite similar to those observed for NH₃ adsorption on

- 1 mesoporous carbon [16], indicating that the response to gas is likely dominated by the nanostructured
- 2 carbon layer in this sample.
- 3



1 **Figure 9:** Resistance versus time during NH₃ gas exposure (a, b); ammonia partial pressure (grey
2 dashed line) and dose (black solid line) versus time (c, d); resistance relative response versus dose
3 (e, f) for bare ZnO NRs and C/ZnO sample, respectively.

4
5 As reported in the literature, type I physisorption isotherms are given by microporous solids [17]. A
6 microporous material, as classified by the IUPAC, is a material containing pores with diameter less
7 than 2 nm. Consistent with the SEM observation of the ZnO NRs reported in Figure 1a, and the
8 adsorption isotherm in Figure 9e, physisorption from a microporous solid adsorbent is occurring.

9 NH₃ molecules physisorb on the ZnO NRs, i.e. they cover the nanorods surface, without being
10 chemisorbed, until saturation is reached. Type VI isotherms show that adsorption isotherms can have
11 one or more steps [33] and represent multilayer adsorption [17]. From Figure 9f we observe the same
12 curve shape of an isotherm of type VI, which leads us to consider that a multilayer adsorption process
13 is occurring. Consistent with the fast XPS data reported in Figure 7, which shows the predominance
14 of peaks related to NH₃ decomposition products (or chemisorption), the isotherm curve in Figure 9f
15 can be interpreted as a type VI isotherm with a multilayer NH₃ adsorption. Having in mind the
16 morphology (Figure 1b) of the hybrid C/ZnO nanostructures, the fast XPS and adsorption isotherms
17 data, the process occurred most likely as following: after the first layer adsorbed on top of the surface,
18 NH₃ molecules decompose, making free space, i.e. new free adsorption sites, for other NH₃ gas to be
19 adsorbed again. These considerations could also justify the higher sensitivity in terms of electrical
20 resistance variation, and it opens up the possibility of using such C/ZnO nanostructures for gas
21 sensing and ammonia storage.

22

23 **4. Conclusions**

24 Fast XPS experiments with synchrotron radiation have been carried out in order to track the
25 interaction of ammonia with hybrid materials based on ZnO nanorods and nanostructured carbon.
26 Direct evidence of NH₃ adsorption on nanostructured hybrid interfaces is provided, combined with

1 transport measurements during the exposure. This operando methodology allowed us to track the
2 adsorption isotherms and relate the different behavior to the sample morphology at the nanoscale.
3 The behaviour of the ZnO nanorods is similar to that observed for physisorption on microporous
4 solids and of the C/ZnO hybrid to chemisorption on mesoporous carbon. While on bare ZnO
5 nanorods, physisorption of ammonia is detected, along with photon beam assisted cracking, on the
6 nanostructured carbon the formation of amine groups is detected, without displaying any effects
7 induced by photon beam exposure. The response $\Delta R/R_0$ of the C/ZnO sample to NH₃ exposure is
8 found to be higher than for the bare ZnO nanorods, suggesting that the C/ZnO hybrid can be regarded
9 as a promising material for gas sensing or for ammonia storage and transportation.

10

11 **Acknowledgements**

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13 STEP program.

14

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