Sustainable Chemistry & Engineering

pubs.acs.org/journal/ascecg

Research Article

Graphite Felt-Sandwiched Ni/SiC Catalysts for the Induction Versus Joule-Heated Sabatier Reaction: Assessing the Catalyst Temperature at the Nanoscale

⁴ Lai Truong-Phuoc,* Cuong Duong-Viet, Giulia Tuci, Andrea Rossin, Jean-Mario Nhut, Walid Baaziz, ⁵ Ovidiu Ersen, Mehdi Arab, Alex Jourdan, Giuliano Giambastiani,* and Cuong Pham-Huu*



6 ABSTRACT: The paper describes a series of graphite felt-7 sandwiched k Ni/SiC composites at variable metal loading (k = 10, 8 15, and 20 wt %) and their application as catalysts for the CO₂ 9 methanation process (Sabatier reaction) under two distinct and 10 conceptually different heating setups: Joule heating versus induction 11 heating (IH). A comparative analysis carried out on all catalysts from 12 this series operated under the two heating configurations has 13 unveiled the superior performance of radiofrequency (RF)-heated 14 (IH) catalysts in the process. Most importantly, it has offered a 15 practical tool to map the gap existing between the macroscopic 16 temperature value measured at the catalyst bed using a remote-17 sensing thermometer (pyrometer) and that (real) of the excited 18 metal nano-objects (Ni NPs) directly engaged in the RF-heated 19 catalytic process. Besides the evident advantages of IH technology



20 applied to the methanation process in terms of process rates (λ) already under nominally low reaction temperatures, the virtual 21 absence of any thermal inertia and the subsequent fast modulation of the temperature at the catalytic bed demonstrates unique 22 features of this heating technology in terms of process safety (cold-reactor walls) and reduction of energy wastes (neither pre- and 23 postcatalyst heating of reagents and products nor that of the whole reactor volume and its peripheral walls).

24 KEYWORDS: induction heating, methanation reaction, Joule heating, temperature of inductively heated nano-objects, SiC-based catalysts

25 INTRODUCTION

²⁶ The development of integrated chemical strategies for the ²⁷ energy conversion and storage is a key priority of our modern ²⁸ society to address a truly renewable energy-based transition.¹ ²⁹ In the power-to-gas (P2G) chain, the exceeding amount of ³⁰ electrical energy produced from renewable and CO₂-free ³¹ energy suppliers (i.e., wind, hydraulic, and solar) is converted ³² into H₂ via water electrolysis. If not immediately used, H₂ can ³³ be employed as a reagent in CO₂ hydrogenation (Sabatier or ³⁴ methanation process)^{2–5} to give synthetic natural gas (SNG; ³⁵ CH₄). SNG is an energy vector of more practical use/ ³⁶ management/distribution within the existing infrastructures ³⁷ and gas grids. Storage and transport conditions for methane are ³⁸ also easier and cheaper compared to those required for ³⁹ hydrogen.

⁴⁰ CO₂ methanation is a highly exothermic process ($\Delta H^0 =$ ⁴¹ -165 kJ mol⁻¹); nevertheless, it requires the use of catalysts ⁴² along with large external heat input to overcome the severe ⁴³ kinetic limitations linked to CO₂ chemical inertness.⁶ External ⁴⁴ heating implies larger energy cost burdening on the process. In ⁴⁵ addition, its combination with the reaction exothermicity and the thermal inertia of classically Joule-heated (JH) reactors 46 hampers an accurate temperature control at the catalytic bed 47 where local temperature gradients (hot spots) can be formed. 48 The latter are typical sources of serious drawbacks and 49 technical limitations linked to the catalyst stability and its 50 performance on long-term runs other than representing a 51 primary energy waste for the process. 52

In recent years, new achievements in catalysis have mainly 53 pointed out on the process intensification⁷ by guaranteeing a 54 quantum leap in its efficiency in terms of time, energy costs, 55 employment of noncritical raw materials, and environmental 56 impact rather than squeezing few percentages on the yield and 57 selectivity of consolidated protocols of the state of the art. The 58 electromagnetic induction heating (IH) or radiofrequency 59

Received: October 23, 2021 Revised: December 9, 2021



138

60 (RF) heating of electrically conductive or magnetic susceptors 61 has already been exploited for a wide range of targeted 62 applications, spanning from biomedical area^{8,9} in the treatment 63 of diseases (through magnetic hyperthermia)¹⁰ to that of 64 processes for metallurgic manufacturing.¹¹ Only recently, this 65 technology has been exploited in catalysis,¹² spotting light on 66 the unique potentiality of this "noncontact" technology for the 67 heat management in highly exo- and endothermic processes.¹² 68 The possibility to convey heat only where it is needed for the 69 chemical process using magnetically or electrically conductive 70 susceptors/catalysts has been used to overcome several heat 71 transfer limitations occurring in classical (Joule) heating 72 schemes. IH indeed limits energy wastes associated to 73 undesired and detrimental heating paths^{12,13} and drastically 74 reduces the thermal inertia (heating/cooling rate) at the 75 catalytic bed.¹²

As far as the CO₂ methanation reaction is concerned, IH has 76 successfully been exploited by means of magnetic nanoparticles 77 (NPs) and/or core-shell systems joining high hyperthermic 78 79 efficiency^{14–16} to excellent catalytic performance. Similarly, 80 electrically conductive susceptors decorated with catalytically 81 active NPs have been employed as robust catalytic systems to 82 run CO₂ hydrogenation under severe and dynamic con-83 ditions.¹² We have recently described an RF-heated methanation scheme based on highly Ni-loaded/ γ -Al₂O₂ (up to 40 wt 84 85 %) composites as electrically conductive and magnetic 86 susceptor/catalysts for rational heat management at the 87 catalytic bed and with a drastic reduction in the nominal temperature values needed to efficiently perform the process 88 $[CO_2 \text{ conversion } (X_{CO_2}) \text{ up to } 98\% \text{ with methane selectivity}$ 89 (S_{CH}) > 99% already at operative temperatures of 210–230 90 °C].¹³ Pieces of evidence from this study have led us to 91 92 conclude that the catalyst temperature measured at the 93 catalytic bed using a remote-sensing thermometer (pyrometer; 94 ø laser beam: \approx 500 μ m, power < 1 mW) is an average value 95 between that of the support (γ -Al₂O₃) and that of the RF-96 excited nickel particles decorating its surface (catalytic sites). 97 These results, together with others related from the 98 literature,^{17,18} support the idea of a temperature gap between 99 the value measured by the laser spot (macroscopic) and that 100 (real) of the heated nano-objects (Ni NPs). Anyhow, the 101 precise determination of the entity of this gap and thus the 102 temperature value at the nanoscale remains a challenging task 103 to be addressed,^{14,19–21} particularly in continuous flow, fixed-104 bed reactors operated under relatively harsh experimental 105 conditions. Current technologies applied to this challenging 106 task are generally based on the development and use of 107 thermosensitive molecules as molecular temperature probes¹⁹ 108 as well as advanced spectroscopic techniques as diagnostic 109 tools for the determination of the local temperature values.²¹ This paper describes the design and synthesis of a new 110 111 family of graphite felt (GF)-sandwiched ^kNi/SiC composites 112 with variable metal loading (k = 10, 15, and 20 wt %) as 113 catalysts/susceptors for the methanation process to be 114 operated under two distinct heating technologies: Joule 115 heating (JH) versus IH. The choice of nickel as a metal active 116 site is based on its renewed catalytic activity and stability in the 117 process. Moreover, its magnetic properties make it a useful 118 susceptor to run the process under IH conditions. SiC is a 119 semiconductor, nonoxide ceramic featured by good thermal 120 conductivity but not suitable to convert electromagnetic 121 energy into heat (not RF-heatable).²² To improve the hyperthermic efficiency of the catalytic system, Ni/SiC was 122 sandwiched between two electrically conductive and IH- 123 responsive (eddy or Foucault currents) GF disks. The 124 comparative analysis of these catalytic materials under different 125 heating configurations (JH vs IH) and experimental conditions 126 [reaction temperature and gas hourly space velocity (GHSV)] 127 has provided a practical tool for mapping the temperature gap 128 existing between the macroscopic value measured at the 129 catalyst bed using the pyrometer and that (real) of the excited 130 metal nano-objects (Ni NPs) directly engaged in the RF- 131 heated catalytic process. 132

At the same time, the catalytic study under IH has pointed 133 out the superior performance of catalysts operated with this 134 direct and noncontact heating technology together with its key 135 advantages in terms of process safety (cold-reactor walls) and 136 reduction of energy wastes.

EXPERIMENTAL SECTION

Materials and Methods. Silicon carbide (SiC) was purchased 139 from Sicat SARL (www.sicatcatalyst.com) as mesoporous pellets $(3 \times 140 \ 1 \text{ mm}, h \times o)^{23-25}$ with a specific surface area (SSA) measured by a 141 N₂ physisorption (at 77 K) of 27 ± 3 m² g⁻¹. SiC pellets were 142 thoroughly washed with distilled water and oven-dried for several 143 hours at 140 °C to remove all powdery fractions prior to be employed 144 as supports for the preparation of the Ni-based composites. Unless 145 otherwise stated, all other reagents and solvents were used as received 146 by the providers without any specific purification/treatment.

Synthesis of ^kNi/SiC Composites. In a typical procedure, SiC 148 pellets (8 g) were impregnated with a proper amount (3.0, 4.8, and 149 6.8 mL) of a 5 M aqueous Ni(NO₃)₂·6H₂O solution to get ^kNi/SiC 150 with a theoretical metal charge of k = 10, 15, and 20 wt %, 151respectively. Impregnated samples were evaporated and oven-dried at 152 110 °C for 2 h before being calcined in air at 350 °C for 2 h as to 153 prepare the corresponding metal oxides. The as-obtained ^kNiO/SiC 154 precatalysts (k = 10, 15, and 20 wt %) underwent reduction treatment 155 under a pure H₂ flow (100 mL min⁻¹) at 350 °C for 2 h. This 156 reduction step was accomplished as an in situ precatalyst treatment 157 whatever may be the nature of the heating setup (JH or IH) employed 158 for the CO₂ methanation trials. Scanning electron microscopy (SEM) 159 was carried out on a ZEISS 2600F instrument with a resolution of 5 160 nm. A 10 kV electron beam was used for SEM imaging operated in 161 the high vacuum mode, using BSE and SE detectors. For 162 measurements, samples were deposited onto a double-face graphite 163 tape holder as to avoid the charging effect during the analysis. 164 Transmission electron microscopy (TEM) was carried out on a JEOL 165 2100F working at 200 kV accelerated voltage, equipped with a probe 166 corrector for spherical aberrations, and a point-to-point resolution of 167 0.2 nm. Elemental mapping was finally accomplished by scanning 168 transmission electron microscopy (STEM) on a JEOL 2100F electron 169 microscope equipped with a Gatan Energy Filter and cold field- 170 emission gun operated at 200 kV with 1.5 Å lattice resolution. For 171 these measurements, samples were dispersed by ultrasound treatment 172 (5 min) in an ethanol solution and a drop of each suspension was 173 deposited on a copper grid covered with a holey carbon membrane for 174 observation. Powder X-ray diffraction (PXRD) measurements were 175 carried out on a D8 ADVANCE Bruker diffractometer with a Cu K α 176 X-ray source ($\lambda = 1.5406$ Å). Rietveld refinements are performed 177 using GSAS-II software.²⁶ The mean size of ordered (crystalline) 178 domains (τ) is calculated from the Debye–Scherrer equation: $\tau = K\lambda/179$ $\beta \cos \theta$, where K is a shape factor, λ is the X-ray wavelength, β is the 180 line broadening at half the maximum intensity (fwhm), and θ is the 181 Bragg angle. The Brunauer-Emmett-Teller (BET) SSAs were 182 calculated from N₂ physisorption isotherms recorded at 77 K on an 183 ASAP 2020 Micromeritics instrument. Pore size distribution was 184 determined by the Barrett-Joyner-Halenda (BJH) method applied 185 to the desorption isotherm branches. Each sample was degassed/ 186 activated at 250 °C under vacuum for 8 h in order to desorb moisture 187 and adsorbed species on its surface. Thermogravimetric analyses 188

189 (TGAs) were run under air (100 mL min⁻¹) on a TGA Q5000 190 thermogravimetric analyzer (TG/DTA) using a heating rate of 10 °C/min. The average weight measured for each sample was 191 192 maintained around 10 mg maximum. Inductively coupled plasma 193 optical emission spectrophotometry (ICP-OES) measurements on mineralized ^kNiO/SiC precatalysts were accomplished on a Varian 194 195 720 ES ICP-OES instrument. The IH setup (EasyHeat 8310, 4.2 kW, 196 Ambrell Ltd) is made of a six-turn spiral induction coil (length = 1.05 197 m, pure coil resistance = $2.066 \times 10^{-3} \Omega$), cooled by means of an 198 external chiller containing water/glycerol (10%, v/v) as the cooling 199 mixture. For the temperature measurement in a typical RF-heated (IH) experiment, temperature at the catalyst bed was monitored/ 200 201 controlled/regulated in almost real-time using a PID system 202 (proportional integral derivative controller, Eurotherm model 3504) 203 connected to a laser pyrometer (Optris, ø laser beam: \approx 500 μ m, 204 power < 1 mW, located at \approx 15 cm from the catalyst and working in 205 the 150–1000 °C range with an accuracy ± 1 °C), shot up on the 206 catalyst/susceptor. A standard calibration procedure²⁷ has been used 207 to fix the emissivity factor for each k Ni/SiC (k = 10, 15, and 20 wt %) 208 catalyst of the study. The heating/cooling rate allowed for the system was about 60-80 °C min⁻¹ in the 150-300 °C temperature range. 209 210 The inductor frequency was constantly maintained at 265 ± 5 kHz, while the current flowing through the induction coils varied from 200 211 212 to 450 A.

IH Setup. In a typical experiment, the quartz reactor containing the tatalyst was housed inside the coils of the induction heater and temperature real-time control/regulation was ensured using a PID system (proportional integral derivative controller, Eurotherm model system) connected to a laser pyrometer (remote-sensing thermometer) shot up on the catalyst bed (see Figure 3 for the sake of clarity).

JH Setup (Electrical Oven). In a typical experiment, the quartz 219 220 reactor containing the catalyst was housed in an electrical oven (ERALY Co., ϕ_{OD} = 200 mm; ϕ_{ID} = 55 mm; depth: 300 mm; I_{max} = 221 222 8.6 A, and T_{max} = 1100 °C). The temperature of the system was 223 monitored using two type-K thermocouples, one for regulating the 224 oven temperature $(T_{\rm F})$ and an additional one located in close contact 225 with the catalytic bed $(T_{\rm C})$ for measuring the temperature swings 226 throughout the process. For these trials, we used thermocouples of 227 ϕ_{ED} . 0.5 mm, very close to the dimension of the laser pyrometer spot 228 in IH. Before each catalytic run, the catalyst was allowed to reach and 229 stabilize (30-45 min) at the target temperature under a pure stream 230 of He (max temperature deviation recorded between $T_{\rm F}$ (F = furnace) 231 and $T_{\rm C}$ (C = catalyst) after stabilization = ± 3 °C) (see Figure 3 for 232 the sake of clarity).

233 Catalytic Tests. The CO₂ methanation reaction (eq 1) was 234 conducted at atmospheric pressure in a fixed-bed quartz tubular 235 reactor ($\phi_{ID} = 12 \text{ mm}$ and length = 400 mm) charged with 0.6 g of the 236 ^kNi/SiC catalyst ($k = 10, 15, \text{ and } 20 \text{ wt }\%; V \approx 0.8 \text{ cm}^3$) and housed 237 on the proper heating setup (EasyHeat 8310 IH setup or a classical 238 external furnace).

$$_{239}$$
 CO₂ + 4H₂ \rightarrow CH₄ + 2H₂O $\Delta H = -165 \text{ kJ mol}^{-1}$ (1)

In a typical catalytic run, a H_2/CO_2 gas mixture (4 v/v) at variable 240 241 GHSVs obtained by a series of calibrated mass flow controllers (Brookhorst) was continuously fed through the catalytic bed 242 maintained at the target temperature. Gases at the reactor outlet 243 244 are passed through a trap filled with silicon carbide pellets where 245 water is condensed before reaching the gas chromatograph for 246 analysis. Reactants and products were analyzed online at the 247 respective reactor outlets using an R3000 (SRA Instrument) microgas chromatograph (μ GC) equipped with an MS5A column for H₂, 248 249 CH₄, and CO detection and a PPU column for CO₂ and C₂ detection 250 with a thermal conductivity detector (TCD). All reactors exit lines 251 were maintained at 110 °C by external heating tapes as to avoid water 252 condensation in the feed.

²⁵³ CO₂ conversion (X_{CO_2}) and CH₄ selectivity (S_{CH_4}) were calculated ²⁵⁴ according to the following equations (eqs 2 and 3)

$$X_{\rm CO_2}(\%) = \frac{F_{\rm CO_2(in)} - F_{\rm CO_2(out)}}{F_{\rm CO_2(in)}} \times 100$$
(2) 255

$$S_{\rm CH_4} (\%) = \frac{F_{\rm CH_4(out)}}{F_{\rm CH_4(out)} - F_{\rm CO(out)}} \times 100$$
(3) 256

where $F_{i(\text{in/out})}$ (mL min⁻¹) is the flow rate of each component in the 257 gas feed at the reactor inlet or outlet. 258

RESULTS AND DISCUSSION 259

Synthesis and Characterization of ^kNi/SiC Catalysts (k ²⁶⁰ = 10, 15, and 20 wt %). ^kNi/SiC composites (k = 10, 15, or ²⁶¹ 20 wt %) were straightforwardly prepared using a wet ²⁶² impregnation technique followed by conventional thermal ²⁶³ calcination/reduction steps (see the Experimental Section for ²⁶⁴ procedure details). The impregnated solids were oven-dried at ²⁶⁵ 110 °C for 2 h and calcined in air at 350 °C for additional 2 h ²⁶⁶ as to convert the nickel nitrate into its corresponding oxide. ²⁶⁷ Catalysts were further reduced under a hydrogen flow (100 mL ²⁶⁸ min⁻¹) at 350 °C for 2 h. All composites from this series were ²⁶⁹ thoroughly characterized by PXRD, TEM/STEM, and N₂- ²⁷⁰ physisorption analyses, while their effective metal loading was ²⁷¹ determined (on NiO precatalysts) through ICP-OES measure- ²⁷² ments.

The catalyst composition and the average Ni NP size before 274 and after the NiO reduction were systematically analyzed 275 through PXRD. Figure 1 shows the PXRD patterns recorded 276 f1



Figure 1. PXRD profiles of 10 NiO/SiC and k Ni/SiC (k = 10, 15, and 20 wt %) for comparison.

for the three $^{k}Ni/SiC$ (k = 10, 15, and 20 wt %) samples in 277 comparison with the model system ¹⁰NiO/SiC, arbitrarily 278 selected as a representative precatalytic system from this series. 279 Hydrogenation (H₂ flow: 100 mL min⁻¹, at 350 °C for 2 h) of 280 all calcinated precatalysts provides a complete NiO conversion 281 into the corresponding Ni⁰ particles. NiO diffraction peaks (at 282 2θ = 37.3, 43.4, 63.0, 75.6, and 79.6°, respectively)²⁸ are 283 almost quantitatively suppressed in all reduced samples, while 284 new distinctive peaks ascribed to Ni⁰ appear at 2θ = 44.5, 51.8, 285 and 76.3°.²⁹ Expectedly, the peak intensity of the latter grows 286 up appreciably when passing from 10 to 20 wt % of metal 287 loading in the SiC normalized spectra. Notably, the chemical 288 inertness of the SiC carrier is likely at the origin of the pure Ni⁰ 289 phases obtained in all samples after the reduction step. The 290 absence of strong chemical interactions between the metallic 291 phase and the nonoxide ceramic support is witnessed by the 292 absence of hardly reducible forms²² at the interface, resistant to 293 294 the relatively mild reduction conditions used. The Ni particle 295 size was roughly estimated using the Scherrer equation³⁰ 296 applied to the peak fwhm of diffraction peaks at $2\theta = 44.5$ and 297 51.8°. The mean values measured for the ^kNi/SiC samples 298 were fixed to 13 ± 1 , 21 ± 2 , and 26 ± 2 nm for k = 10, 15, 299 and 20 wt %, respectively.

The higher the metal loading, the higher is the mean nickel 300 301 particle size and the generation of larger metal aggregates. 302 While Ni/SiC composites at relatively low Ni contents (i.e., 303 ¹⁰Ni/SiC) can boast a higher stability extent due to the 304 presence of amorphous SiO_2/SiO_xC_y layers at the carrier 305 topmost surface that contribute to stabilize and disperse the 306 metal active phase,^{22,31} higher metal loadings and larger metal 307 aggregates on supports with moderate SSA are less leaching-308 resistant. As an example, SEM images of the ²⁰Ni/SiC catalyst 309 (Figure S1) at different magnifications unveil the almost 310 complete coating of the SiC carrier with nickel superstructures 311 and structural vacancies; the latter are probably originated by 312 the partial removal of poorly stabilized nickel aggregates. 313 Therefore, the higher the metal content used in the catalysts' 314 preparation, the higher the deviation between the theoretical 315 and the effective metal loading in the composites. This trend 316 was confirmed by the ICP-OES analysis on the final catalysts. 317 While the ¹⁰Ni/SiC sample showed an excellent match 318 between the theoretical and the measured nickel loading (9.3 $_{319} \pm 0.3$ wt %), ²⁰Ni/SiC displayed a higher deviation from the 320 theoretical content (17.0 \pm 0.8 wt %) (Table 1). At odds with

Table 1. ICP-OES Analysis, SSA, Pore Volume, and Pore Size Distribution Measured on Each ^kNi/SiC (k = 10, 15, and 20 wt %) Composite in Comparison with the Bare Ceramic Support

entry	sample	Ni wt % (ICP-OES)	SSA^{a} $(m^{2} g^{-1})$	total pore volume ^b (cm ³ g ⁻¹)	average pore size ^c (nm)
1	SiC		27	0.16	23.8
2	¹⁰ Ni/SiC	9.3 ± 0.3	29	0.14	17.6
3	¹⁵ Ni/SiC	14.0 ± 0.5	26	0.13	21.6
4	²⁰ Ni/SiC	17.0 ± 0.8	24	0.13	22.6
^a bft	SSA meas	ured at $T =$	77 K ^b Tot	al nore volum	e determined

"BET SSA measured at T = 77 K. "Total pore volume determined using the adsorption branch of the N₂ isotherm at $P/P_0 = 0.98$. "Determined by BJH desorption average pore width (4V/A).

321 XRD data, XPS surface analyses of all Ni/SiC composites 322 unveiled the presence of NiO_x forms (Figure S2A,B)³² 323 resulting from a rapid passivation of metal deposits during 324 the sample handling in air and their manipulation for analysis 325 (formation of thin metal-oxide layers). For this reason, each 326 NiO/SiC precatalyst underwent H₂ flow reduction at 350 °C 327 in situ (see the Experimental Section) just prior of its exposure 328 to the reagents' mixture for catalysis. The SSA, the total pore 329 volume, and the mean pore size measured for all Ni composites (Table 1, entries 2-4 and Figure S3B-D) were very similar 330 331 and close to the morphological parameters recorded for the 332 bare SiC support (Table 1, entry 1 and Figure S3A). All 333 samples present classical type-II isothermal profiles and 334 moderate SSA values. The meso-macroporous nature of the 335 ceramic carrier was moderately affected by the deposition of Ni 336 NPs and by the final metal loading in the composite. Although 337 any conclusion based on these moderate morphological 338 deviations remains speculative and poorly significant to the

manuscript purposes, it can be simply claimed that the higher 339 the metal charge, the lower is the SSA value of the composite. 340

TEM on ^kNi/SiC (Figure 2A,B for k = 10 and Figure 2D,E 341 f2 for k = 15 wt %) showed a relatively homogeneous dispersion 342 of metal NPs all over the SiC support. Moreover, a statistical 343 analysis of NP size distribution carried out on each catalyst 344 (Figure 2C,F) was in good agreement with PXRD outcomes. 345 High-resolution (HR) TEM analysis (Figure S4) on ¹⁰Ni/SiC 346 has finally revealed the presence of crystalline phases featured 347 by distinctive interplanar distances of 0.21 and 0.24 nm, 348 characteristic of the (1 1 1) planes of Ni³³ and SiC³⁴ phases, 349 respectively. 350

Methanation Reaction with ^kNi/SiC as a Catalyst $_{351}$ under Two Heating Configurations: JH Versus IH. $_{352}$ Looking for more efficient and energy-saving approaches to $_{353}$ the methanation reaction, the ^kNi/SiC (k = 10, 15, and 20 wt $_{354}$ %) catalysts were investigated under two different reactor $_{355}$ configurations: (i) a classical reactor scheme (Figure 3A) using $_{356}$ $_{63}$ an electrical furnace for the catalyst heating through $_{357}$ convection, conduction, and/or radiation heat transfer $_{358}$ (named hereafter as Joule-heating mode; JH) or (ii) an IH $_{359}$ setup (Figure 3C) based on the electromagnetic properties of a $_{360}$ susceptible medium (susceptor) exposed to a varying magnetic $_{361}$ field (H) and its ability to convert the electromagnetic energy $_{362}$ into heat at the catalytic bed or neighboring it (named $_{363}$ hereafter induction-heating mode; IH).

For both reactors, temperature values were registered at the 365 respective macroscopic heating sources. Therefore, for a JH 366 system (Figure 3A), the temperature was monitored using two 367 thermocouples housed in the furnace (temperature of the 368 furnace, T_F) and in close contact with the quartz reactor 369 (temperature of the catalyst, T_C), respectively. For the RF- 370 heated setup (Figure 3C), the temperature was measured using 371 a remote-sensing thermometer (pyrometer) with its laser beam 372 directly shot over the GF susceptor (see the Experimental 373 Section and Section 3.4 for details).

Whatever may be the heating configuration employed, $^{k}Ni/_{375}$ SiC powders were sandwiched in the quartz reactor between 376 two GF disks (Figure 3B). The latter were innocent, porous, 377 and thermal conductor supports for the catalytic material when 378 the reaction was operated under the JH configuration (Figure 379 3A), whereas they exhibited the role of susceptors for the 380 electromagnetic energy conversion into heat when the reaction 381 was carried out under IH (Figure 3C; also see Section 3.4 for 382 details). Blank tests carried out using the GF alone did not 383 show any CO₂ methanation activity in the range of operational 384 temperatures. 385

Effect of the Reaction Temperature and Heat ³⁸⁶ Management Setup (JH vs IH) on the Methanation ³⁸⁷ Performance. In a first set of experiments, we studied the ³⁸⁸ effect of the reaction temperature at the catalyst bed on its ³⁸⁹ performance using both heating configurations (JH vs IH), ³⁹⁰ while maintaining the amount of catalyst (600 mg) and the ³⁹¹ reagent GHSV (10,000 mL g⁻¹ h⁻¹) constant. As Figure 4 ³⁹² f4 shows, catalysis under JH (blue - \bullet -) presented classical ³⁹³ sigmoidal trends for CO₂ (X_{CO_2}) conversion into SNG, ³⁹⁴ whatever may be the catalytic system at work (^kNi/SiC, k =³⁹⁵ 10, 15, and 20 wt %). Although ¹⁰Ni/SiC reached the higher ³⁹⁶ CO₂ conversion (X_{CO_2}) at 350 °C (and kept it almost ³⁹⁷ unchanged in the 350–375 °C range), ¹⁵Ni/SiC and ²⁰Ni/SiC ³⁹⁸ showed their higher performance at 375 °C. As expected, the ³⁹⁹ higher the nickel loading, the higher is the X_{CO_2} although the ⁴⁰⁰



Figure 2. TEM images recorded on 10 Ni/SiC (A,B) and 15 Ni/SiC (D,E) at different magnifications along with the statistical distribution of NP sizes determined over 100 metal particles for each sample. (C,F) Ni NP size distribution on 10 Ni/SiC and 15 Ni/SiC, respectively.



Figure 3. Representation of the adopted reactor configurations to carry out the methanation process; (A) electrical furnace (hot-wall reactor) operating with a classical JH scheme based on heat convection, conduction, and radiation; (B) details of the kNi/SiC catalyst sandwiched between two GF disks (susceptors) and housed in the quartz tube reactor; and (C) inductor coil for the RF heating of a susceptible medium (GF, susceptor) exposed to an external varying magnetic field (H). Thermal spectra reported below each reactor configuration account for the classical temperature diffusion (to and from the catalyst) depending on the operating heating technology (JH or contact and IH or contactless).

⁴⁰¹ specific catalyst rate (λ) expressed as mol_{CH4} produced per g_{Ni} ⁴⁰² per *h* was superior with ¹⁰Ni/SiC (see Table S1).

⁴⁰³ Under these operative temperatures (<400 °C) and ⁴⁰⁴ irrespective of the employed heating setup, the SNG process ⁴⁰⁵ selectivity laid constantly close to 100% with no traces of CO ⁴⁰⁶ or other byproducts detected at the reactors' outlet. The ⁴⁰⁷ remarkably high S_{CH_4} was partially due to the thermal ⁴⁰⁸ conductivity of the SiC carrier²² that was thought to mitigate the generation of local temperature gradients (hot spots) by 409 diffusing extra heat to the whole catalyst volume³⁵ and 410 ultimately favoring its removal by the action of the gaseous 411 reagent stream (solid–gas heat exchange). 412

Such a heat exchange was even more efficient in a RF-heated 413 catalyst because of the absence of reagents flow preheating 414 until they came in contact with the (hot) catalyst surface. 415

When CO_2 methanation was carried out under IH (red 416 •), we observed an appreciably higher catalyst performance 417



Figure 4. CO_2 methanation with ^kNi/SiC [k = 10 (A), 15 (B), and 20 wt % (C)] as a function of the reaction temperature. All catalysts were tested without any thermic diluent. Reaction conditions: catalyst weight = 600 mg, GHSV (STP) = 10,000 mL g⁻¹ h⁻¹, [CO_2] = 20 vol %, [H_2] = 80 vol %, H_2 -to- CO_2 ratio = 4, total flow rate = 100 mL min⁻¹, atmospheric pressure. Red (- \bullet -) and blue (- \bullet -) curves refer to the X_{CO_2} vs temperature as experimentally determined within an IH and JH setup, respectively. CH₄ selectivity (S_{CH_4} , not shown in figures) was constantly equal to 100% in the whole temperature range. Thermodynamic equilibrium conversion (dashed black line "-----") has been included for the sake of comparison. All experimental data have been fitted with sigmoidal logistic functions of type I.



Figure 5. Arrhenius plots of ^kNi/SiC (k = 10, 15, and 20 wt %) in CO₂ methanation under the classical JH (A) or IH (B) setup. λ is expressed as mol_{CH4} g_{Ni}⁻¹ h⁻¹ and measured in the temperature range where X_{CO} , reaches 30% conversion maximum.

 $_{418}$ already for temperatures below 350 $^\circ\text{C}\textsc{,}$ whatever may be the metal loading of the catalyst at work. Remarkably, X_{CO_2} 419 420 measured at 280 and 300 °C with ¹⁵Ni/SiC was up to 48 and 43% higher than that measured under the same 421 422 experimental conditions but for catalysts operated under the 423 JH mode (Figure 4B). As a result, higher productivity values (λ) were recorded for the RF-heated ¹⁵Ni/SiC already at 424 nominal low temperature (280 °C) values (Table S1). This 425 426 result is of great relevance particularly in light of an industrial 427 exploitation of these catalysts in combination with IH 428 technology. The higher catalyst specific rate (λ) measured at 429 low reaction temperatures with the RF-heated ¹⁵Ni/SiC fulfils

at least three key prerequisites of a sustainable methanation $_{430}$ scheme: (i) catalysts with a relatively low loading of a highly $_{431}$ dispersed and noncritical metal active phase; (ii) relatively $_{432}$ mild operative temperatures to get high $X_{\rm CO_2}$ values chemo- $_{433}$ selectively into SNG; and (iii) a safer and really energy-saving $_{434}$ reactor configuration for the heat management at the catalytic $_{435}$ bed.

It should be stressed that Ni composites at higher metal 437 loading (i.e., ¹⁵Ni/SiC and ²⁰Ni/SiC) deviate appreciably from 438 the classical sigmoidal profile when catalysis was operated 439 under IH. Such a behavior was ascribed to a more complex 440 heat management at the catalytic bed in the case of RF-heated 441

Table 2. Nickel Particle Size in the Three ^kNi/SiC (k = 10, 15, and 20 wt %) and Activation Energy Values (E_a) Calculated for the CO₂ Methanation Reaction Carried Out under JH or IH from the Respective Arrhenius Plots

entry	sample	Ni NPs ø (nm) ^a	E_{a} (JH) ^b (kJ mol ⁻¹)	E_{a} (IH) ^b (kJ mol ⁻¹)	ΔE_{a} (JH – IH) (kJ mol ⁻¹)	$\Delta T_{\rm max} (^{\circ}{\rm C})^{c}$	$M_{\rm s} ({\rm emu/g})^d$
1	¹⁰ Ni/SiC	13 ± 1	75.0	72.2	2.8	26 ± 3	47.5 ± 0.7
2	¹⁵ Ni/SiC	21 ± 2	76.0	64.6	11.4	81 ± 4	51.2 ± 0.6
3	²⁰ Ni/SiC	26 ± 2	76.0	39.5	36.5	95 ± 4	52.4 ± 0.4

^{*a*}Determined by XRD from the Scherrer equation and the peak fwhm of the diffraction peaks at $2\theta = 44.5$ and 51.8° . ^{*b*}Calculated from the curves' slopes of Arrhenius plots in the temperature range where X_{CO_2} reaches 30% conversion maximum. ^{*c*}Determined as the maximum difference on the *x*-axis (ΔT , ^oC) of curves in Figure 4A,C between two points at the same CO₂ conversion (X_{CO_2} on JH and IH curves) as the maximum ΔT value (ΔT_{max}) between the GF temperature (bulk temperature) and that effectively reached at the excited metal catalyst particle sites. ^{*d*}Specific saturation magnetization (M_s) determined for the different Ni particle sizes on the basis of eq 4 and assuming a mean NiO layer of 0.5 nm in accordance with the TEM analysis on the samples of this study.

442 catalysts. Indeed, contributions to the catalyst heating derive in 443 part from the inductively heated GF disks (heat transfer by 444 convection/conduction to the thermally conductive SiC 445 support), in part from dissipation paths related to the 446 electrically conductive and magnetic nature of the large-sized 447 Ni NPs (i.e., Eddy currents or Foucault currents and/or heat 448 supplied via hysteresis losses).¹² As a proof of evidence, lower 449 X_{CO_2} and λ values measured under IH for temperatures higher 450 than 350 °C (Figure 4B,C) can be reasonably attributed to the 451 loss of one heat dissipation path (e.g., heat hysteresis) as a 452 function of the temperature effect on the metal magnetic 453 properties (vide infra Section 3.4 for details).^{36,37}

Arrhenius plots (Figure 5) constructed as the logarithm of 454 455 the catalyst process rate $[Ln(\lambda)]$ versus the reciprocal of the 456 reaction temperature (K⁻¹) have contributed to better 457 distinguish among the catalytic performance of ^kNi/SiC 458 catalysts operated under the two heating setups (Figure 3). 459 Expectedly, when CO₂ methanation was carried out under the 460 JH mode, the three catalysts behave similarly from a kinetic 461 viewpoint, showing almost equal activation energy (curves' 462 slope in Figure 5A and E_a values in Table 2) for processes $_{463}$ operated under identical conditions. Moreover, calculated E_a 464 values were in excellent agreement with the related literature 465 data for the methanation process.³⁸⁻⁴⁰ Reversely, IH showed 466 important deviations in the curves' slope (Figure 5B), hence 467 implying relevant alterations of the activation energy values associated to the process. 468

f5

As Table 2 shows, the higher the catalyst metal loading, the 470 lower is the activation energy (E_a) for the inductively heated 471 methanation process and thus the higher is the $\Delta E_a(JH - IH)$ 472 gap measured for the reaction operated under different heating 473 configurations.

It is evident that differences in the activation energy values 475 (ΔE_a) measured for the same process, operated with the same 476 catalyst but under different heating schemes (JH vs IH), had to 477 be ascribed to the existence of a temperature gap between the 478 value measured (macroscopically) at the GF surface and those 479 (real) of the RF-heated nano-objects (nickel active sites) at the 480 SiC surface.

481 Several precedents from the literature have already 482 demonstrated the existence of important temperature discrep-483 ancies (up to orders of magnitude)¹⁷ between values reached 484 at the surface of RF-heated nano-objects (metal NPs) and 485 those measured macroscopically using a remote-sensing 486 thermometer on the bulk materials.^{18,41-43} The temperature 487 measurement at the level of single-catalyst particles (especially 488 for magnetically responsive elements) remains a challenging 489 issue to be addressed, and it becomes even more tricky under operando conditions in severe experimental environ- 490 ments.^{14,19–21} Anyhow, from the analysis of curves in Figure 491 4A–C, it can be inferred that the maximum difference on the 492 x-axis (ΔT , °C) between two points at the same CO₂ 493 conversion (X_{CO_2} on JH and IH curves) corresponds to the 494 maximum ΔT value (ΔT_{max}) between the "macroscopic" 495 temperature measured at the GF disks and that actually 496 reached at the RF-excited nickel particles. Accordingly, 26 ± 3, 497 81 ± 4, and 95 ± 4 °C are the ΔT_{max} values between GF and 498 Ni NPs in the RF-heated Ni/SiC catalysts containing 10, 15, 499 and 20 wt % of metal NPs, respectively.

As expected, such a ΔT value varies as a function of the 501 magnetic properties of Ni NPs, hence their mean particle size 502 as well as the dependence of the nickel specific saturation 503 magnetization ($M_{\rm s}$) from the temperature (Section 3.4 for 504 details). 505

On the Origin of ΔT Values in RF-Heated ^kNi/SiC ⁵⁰⁶ Catalysts Applied to the CO₂ Methanation Reaction. ⁵⁰⁷ There are no doubts that the two heating configurations (JH ⁵⁰⁸ and IH) hold distinct thermal and energetic features. Indeed, ⁵⁰⁹ the RF heating of a GF-sandwiched catalyst always occurs in a ⁵¹⁰ "cold-wall reactor" modality,²⁰ hence avoiding any pre- and ⁵¹¹ postheating of reagents and products, including that of the ⁵¹² whole reactor volume and its peripheral walls. Such a ⁵¹³ configuration fosters (among the others) the reaction ⁵¹⁴ thermodynamics by favoring H₂O condensation just after the ⁵¹⁵ (hot) catalytic bed and reduces all thermal inertia phenomena ⁵¹⁶ typically encountered on furnace-based reactors while ⁵¹⁷ guarantying more sustainable and energy-saving catalytic ⁵¹⁸ schemes.¹²

Similar benefits linked to a more convenient heat manage- 520 ment at the catalytic reactor have recently been commented by 521 others for different gas-phase processes. 44,45 522

It is clear that classical heat transfer by conduction/radiation 523 (JH) applies to both heating configurations (JH and IH) 524 proposed in this work. Under IH, the GF-sandwiched catalyst 525 realizes a sort of micro-JH reactor. The thermal dissipation 526 caused by the electromagnetically induced eddy currents 527 flowing through the GF fibers can be considered as the main 528 source of external heating (Figure 3B) for the catalyst active 529 sites via classical conduction/radiation heat exchange. As an 530 electrical semiconductor, SiC is not suitable to directly convert 531 electromagnetic energy into heat at the catalytic bed; however, 532 its thermal conductivity conveys the heat produced by the GF 533 disks to the catalyst-active sites. Assuming the methanation 534 exothermicity as independent from the nature of the adopted 535 heating setup, under IH, the electrically conductive and 536 magnetic nickel deposits (catalyst active phase) undergo 537



Figure 6. (A) X_{CO_2} and productivity rate (λ) dependence from the applied GHSV (L g⁻¹ h⁻¹) for methanation processes operated on ¹⁵Ni/SiC as a catalyst under the JH (round symbols) and IH (square symbols) mode. (B) Long-term methanation test at 20,000 mL g⁻¹ h⁻¹ as GHSV under the discontinuous mode using the inductively heated ¹⁵Ni/SiC catalyst. Each break corresponds to the reactor switching-off overnight. Other common reactional conditions for A and B: $[CO_2] = 20$ vol %; $[H_2] = 80$ vol %; H_2 -to-CO₂ ratio = 4; catalyst weight = 0.6 g; reaction temperature = 300 °C, atmospheric pressure.

538 dedicated heating paths.⁴⁶ In particular, eddy currents (or 539 Foucault currents)^{47,48} flowing through the larger metal 540 aggregates and hysteresis loss^{49,50} phenomena can induce 541 local heating directly at the metallic phase. Hysteresis loss 542 depends from the specific saturation magnetization (M_s) of the 543 metal particles. It is a key property of ferromagnetic particles and varies as a function of the metal NP size and their ultimate 544 temperature. Kuz'min demonstrated with a simple analytical 545 546 representation the shape of the temperature dependence⁵¹ for 547 bulk Ni NPs. He properly described the finite-temperature properties of these ferromagnets in the -273.15 °C Curie 548 549 temperature $(T_c, {}^{\circ}C; {}^{Ni}T_c = 355 {}^{\circ}C)^{52}$ range, with the latter s50 value corresponding to the temperature where M_s vanishes.³⁶ 551 Duan and co-workers finally combined the morphological 552 properties (TEM and XRD) of various grain-sized Ni NPs 553 prepared by the autocatalytic reduction of a Ni^{II} salt with the $_{554}$ $M_{\rm s}$ size dependence. Their study provided a useful model for sss the calculation of the experimental M_s value for similarly 556 prepared Ni NP-based catalysts in the form of spontaneously 557 surface-passivated (NiO-coated) systems like ours (see Section 3.1 for details). Based on their model (eq 4) and assuming a 558 559 mean size of our Ni NPs of 13, 21, and 26 nm with an average 560 thickness of the NiO thin layer for all catalysts of about 0.5 nm $_{561}$ (see also Figure S4B), we calculated $M_{\rm s}$ values comprised 562 between 47.5 and 52.4 emu/g (see Table 2), in excellent s63 accordance with the Ni NP size/ M_s dependence reported in 564 the literature.

$$M_{\rm s}^{\rm exp} = \frac{\sigma_{\rm s}^{\rm bulk} \times \frac{4}{3}\pi r^3 \times \rho_{\rm Ni}}{\frac{4}{3}\pi r^3 \times \rho_{\rm Ni} + \frac{4}{3}\pi (R^3 - r^3)\rho_{\rm NiO}}$$
(4)

566 where $\rho_{\text{Ni}} = 8.8 \text{ g cm}^{-3}$, $\rho_{\text{NiO}} = 6.827 \text{ g cm}^{-3}$, $\sigma_{\text{s}}^{\text{bulk}} = 57.50$ 567 emu/g, R = mean radius of Ni NPs from XRD analysis, and r = 568 (R – average thickness of a NiO thin layer).

56

The higher the $M_{\rm s}$ value, the higher is the hyperthermic s70 efficiency or specific absorption rate of the nickel-based sample s71 and thus the higher is its capacity to act as a heat susceptor⁵³ s72 once immersed in an AC magnetic field.⁵⁴ The increasing s73 values of $\Delta T_{\rm max}$ measured on Ni/SiC samples with variable s74 nickel-loading and nickel particle size (Figure 4A–C) are s75 perfectly in line with these conclusions. Interestingly, JH and s76 IH curves recorded in Figure 4A–C present similar converging s77 trends in close correspondence to the ^{Ni}T_c value (355 °C). s78 These trends led us to conclude that the extra heat (ΔT) generated under IH at the catalytic nickel sites was essentially 579 due to hysteresis loss dissipation phenomena that were 580 definitively suppressed when the metal sites became para- 581 magnetic, 55,56 that is, above ${}^{Ni}T_{c}$. Accordingly, it can be 582 concluded that eddy current contributions are almost 583 negligible, whatever may be the nickel loading in the three 584 catalysts in comparison. The moderate or null contribution 585 from eddy currents was finally confirmed by the increasing 586 $X_{\rm CO_2}$ conversion gap (or λ gap) measured with the two ^kNi/₅₈₇ SiC catalysts at higher metal loadings (k = 15 and 20 wt %) 588 operated under JH and IH at temperatures >355 °C (above 589 $^{
m Ni}T_{
m c}$). Under these conditions, the higher the catalyst particle 590 size, the higher is the X_{CO_2} conversion gap between the two ₅₉₁ processes. Because eddy currents are directly proportional to 592 the square of the NP radius, their contribution should follow 593 an opposite trend to that recorded for catalysts ¹⁵Ni/SiC and 594 20 Ni/SiC operated in CO₂ methanation at temperatures >^{Ni}T_c. ₅₉₅

According to the observed trends of X_{CO_2} versus temperature ₅₉₆ recorded where the Ni NPs become paramagnetic (yellow 597 sections of Figure 4A-C), it can be concluded that catalyst 598 heating occurred through convection/conduction only. For the 599 two RF-heated catalysts at higher Ni loading (¹⁵Ni/SiC and 600 ²⁰Ni/SiC), the reduced catalyst process rates for temperatures ₆₀₁ > ^{Ni} T_c are then ascribed to a less-effective JH transfer in the 602 micro-JH reactor (GF-sandwiched catalyst) with respect to the 603 classical external furnace. In contrast with the classical furnace, 604 the formally cold walls of the micro-JH reactor facilitate a more 605 rapid heat dissipation from the catalytic bed. Therefore, it can 606 be inferred that this phenomenon in combination with the 607 suppressed magnetic properties of Ni particles further 608 contributed to the observed decrease in the catalysts' 609 performance. 610

Effect of the Reactant Flow Rate (GHSV) on the 611 Methanation Performance of a JH or IH ¹⁵Ni/SiC 612 Catalyst. The influence of the reactant space velocity 613 (GHSV) was deliberately investigated on ¹⁵Ni/SiC as one of 614 the most representative samples from this catalyst series at a 615 temperature of 300 °C where the ΔX_{CO_2} and $\Delta \lambda$ values 616 measured under the two heating configurations (JH and IH) 617 were the highest (Figure 4B and Table S1, entries 15–16). As 618 Figure 6A shows, selectivity toward CH₄ laid constantly over 619 f6 99% with no detectable traces of any other reaction byproduct, 620 whatever may be the adopted heating configuration and the gas 621 $_{622}$ flow applied to the reactor in the 10–25 L $g^{-1}\ h^{-1}$ GHSV $_{623}$ range.

As far as X_{CO_2} and λ values are concerned, the two heating 624 625 configurations translated into radically different catalyst $_{626}$ behaviors. Under the JH mode, X_{CO_2} decreased for increasing 627 flow rates and catalyst productivity (λ) remained substantially 628 unchanged throughout the last three reagent rates. When the ₆₂₉ process was operated under IH, an initial X_{CO_2} increase was 630 observed and the catalyst productivity grew constantly (and 631 appreciably) for increasing reactant flows. It should be stressed 632 that $\Delta \lambda$ in ¹⁵Ni/SiC operated under the two heating schemes 633 (JH and IH) and at increasing reactant flow rates (from 10,000 634 to 25,000 as GHSV) increased over 190% (Table S1, entries 635 15-16 vs 21-22). There were few doubts that different values $_{636}$ and trends in X_{CO_2} measured within the two heating 637 configurations at increasing GHSVs reflected a radically 638 different control of the temperature at the catalytic bed of $_{639}$ the two reactors. All X_{CO_2} decreasing trends measured at 640 increasing GHSVs can be claimed as the consequence of 641 several factors such as: (i) an oversaturation of the metal active 642 sites; (ii) a reduced contact time of reagents with the catalyst 643 active phase; and (iii) a reduced catalyst temperature due to a 644 more effective solid-gas heat exchange at increasing reactant 645 flow rates. All these phenomena may contribute to reducing 646 X_{CO2} when GHSVs increase. However, i and ii items certainly $_{\rm 647}$ do not justify the growing trend of $X_{\rm CO_2}$ measured under IH 648 when the reaction was operated between 10,000 and 15,000 649 GHSVs. In contrast to classical JH reactors, temperature 650 swings at the catalytic bed caused by growing flow rates 651 (warming) and increased solid-gas heat exchanges (cooling) 652 are rapidly controlled and compensated (with almost no 653 thermal inertia) by a fast modulation of the current flowing 654 through the inductor coils (remote-sensing thermometer 655 connected to the inductor through a PID controller). This excellent catalyst temperature control is also at the 656

⁶⁵⁶ Fins excellent catalyst temperature control is also at the ⁶⁵⁷ origin of a prolonged stability and durability of inductively ⁶⁵⁸ heated catalysts on methanation run. Figure 6B refers to a ⁶⁵⁹ long-term methanation test operated with the RF-heated ¹⁵Ni/ ⁶⁶⁰ SiC catalyst at 300 °C with 20,000 mL g⁻¹ h⁻¹ as GHSV and in ⁶⁶¹ a discontinuous process (successive reactor shut-downs; ⁶⁶² breaks) as to add a further stress factor to the long catalytic ⁶⁶³ trial (>25 h). As it can be seen, after an initial stabilization, the ⁶⁶⁴ catalyst constantly laid on an average $X_{CO_2} = 75 \pm 2\%$ with a ⁶⁶⁵ quantitative methane selectivity and a productivity (λ) that ⁶⁶⁶ closely approached 0.9 mol_{CH4} g_{Ni}⁻¹ h⁻¹.

The combination of the IH setup with the inherent thermal conductivity of SiC supports reduces all classical drawbacks sociated to catalyst deactivation on run. TEM analysis of the catalytic material before and after the long-term run has demonstrated the superior stability of the catalytic system once roce distribution (sintering) nor appreciable catalyst coking phenomena after long-term runs were revealed (Figures S5 and ros S6).

676 CONCLUSIONS

677 In summary, we compared ^kNi/SiC composites at variable 678 metal loading as catalysts for the Sabatier process under two 679 different heating configurations (JH vs IH). The study has 680 unambiguously demonstrated the superior performance, stability, and durability of catalysts operated under the less- 681 conventional IH setup. The superior performance of RF- 682 heated catalysts (already at relatively low temperatures) 683 basically stems from a different heat management at the 684 catalytic bed. The comparative analysis between the two 685 heating schemes along with the study of process kinetics has 686 offered a practical and simple tool to the estimation of the 687 temperature gap between that measured directly at the main 688 heat source and that (real) of the RF-heated nano-objects 689 (metal NPs). If we assume any catalyst heating/cooling 690 contributions (i.e., reaction exothermicity; solid-gas heat 691 exchanges at varying GHSVs; and heat dissipation by the 692 SiC support) as equal regardless of the nature of the heating 693 configuration at work (IH and JH), such a temperature 694 difference is unambiguously associated to the distinctive heat 695 dissipation properties (eddy currents and hysteresis loss) of the 696 electrically conductive and magnetic nickel particles immersed 697 in the AC magnetic field.

Besides the evident benefits of the IH technology applied to 699 the methanation process in terms of $X_{\rm CO_2}$ and process rates (λ) $_{700}$ already under nominally low reaction temperatures, the virtual 701 absence of thermal inertia phenomena and the subsequent fast 702 modulation of the temperature directly at the catalytic bed 703 hold unique advantages in terms of process safety (cold-reactor 704 walls) and reduction of energy wastes (neither pre- and 705 postcatalyst heating of reagents and products nor that of the 706 whole reactor volume and its peripheral walls). Finally, the 707 choice of silicon carbide as a support for the metal active phase 708 has guaranteed a better temperature control at the catalytic 709 bed. Indeed, SiC thermal conductivity allows the bidirectional 710 heat transfer between the GF disks (susceptors under IH) and 711 the Ni NPs, avoiding the formation of local hot spots that 712 reduce process selectivity and the catalyst lifetime. 713

ASSOCIATED CONTENT

1 Supporting Information

714 715

724

725

The Supporting Information is available free of charge at 716 https://pubs.acs.org/doi/10.1021/acssuschemeng.1c07217. 717

SEM, HRTEM, BET, and pore size distribution (BJH) 718 of ^{*k*}Ni/SiC (k = 10, 15, and 20 wt %); XPS survey and 719 Ni 2p_{3/2} of ¹⁵Ni/SiC; and X_{CO_2} and catalyst specific rate 720 (λ) for CO₂ methanation with ^{*k*}Ni/SiC (k = 10, 15, and 721 20 wt %) as catalyst(s) at variable reaction temperatures, 722 operated under the JH or IH mode (PDF) 723

AUTHOR INFORMATION

Corresponding Authors

- Lai Truong-Phuoc Institute of Chemistry and Processes for 726 Energy, Environment and Health (ICPEES), ECPM, UMR 727 7515 of the CNRS and University of Strasbourg, 67087 728 Strasbourg Cedex 02, France; Email: ltruongphuoc@ 729 unistra.fr 730 Giuliano Giambastiani – Institute of Chemistry and Processes 731
- for Energy, Environment and Health (ICPEES), ECPM, 732 UMR 7515 of the CNRS and University of Strasbourg, 733 67087 Strasbourg Cedex 02, France; Institute of Chemistry of 734 OrganoMetallic Compounds, ICCOM-CNR and Consorzio 735 INSTM, 50019 Florence, Italy; • orcid.org/0000-0002-0315-3286; Email: giambastiani@unistra.fr, 737 giuliano.giambastiani@iccom.cnr.it 738
- Cuong Pham-Huu Institute of Chemistry and Processes for 739 Energy, Environment and Health (ICPEES), ECPM, UMR 740

- 7515 of the CNRS and University of Strasbourg, 67087 741
- Strasbourg Cedex 02, France; Email: cuong.pham-huu@ 742
- unistra.fr 743

744 Authors

- **Cuong Duong-Viet** Institute of Chemistry and Processes for 745
- Energy, Environment and Health (ICPEES), ECPM, UMR 746
- 7515 of the CNRS and University of Strasbourg, 67087 747 Strasbourg Cedex 02, France 748
- 749

Giulia Tuci – Institute of Chemistry of OrganoMetallic Compounds, ICCOM-CNR and Consorzio INSTM, 50019 750

- *Florence, Italy;* [®] orcid.org/0000-0002-3411-989X 751
- Andrea Rossin Institute of Chemistry of OrganoMetallic 752 Compounds, ICCOM-CNR and Consorzio INSTM, 50019 753
- 754 Florence, Italy

Jean-Mario Nhut – Institute of Chemistry and Processes for 755

Energy, Environment and Health (ICPEES), ECPM, UMR 756

- 7515 of the CNRS and University of Strasbourg, 67087 757
- Strasbourg Cedex 02, France 758

Walid Baaziz – Institute of Physic and Chemistry of Materials 759 of Strasbourg (IPCMS), UMR 7504 CNRS-University of 760

- Strasbourg, 67037 Strasbourg Cedex 08, France 761
- **Ovidiu Ersen** Institute of Physic and Chemistry of Materials 762
- of Strasbourg (IPCMS), UMR 7504 CNRS-University of 763
- Strasbourg, 67037 Strasbourg Cedex 08, France; 764
- orcid.org/0000-0002-1553-0915 765
- Mehdi Arab ORANO Tricastin, Direction de La Recherche 766 & Développement, 26701 Pierrelatte Cedex, France 767
- Alex Jourdan ORANO Tricastin, Direction de La Recherche 768
- 769 & Développement, 26701 Pierrelatte Cedex, France

770 Complete contact information is available at:

771 https://pubs.acs.org/10.1021/acssuschemeng.1c07217

772 Notes

773 The authors declare no competing financial interest.

774 **ACKNOWLEDGMENTS**

775 The present work was supported by ORANO Co. through a 776 contract no. 40104671. G.G. and C.P.-H. would like to thank 777 the TRAINER project (Catalysts for Transition to Renewable 778 Energy Future) of the "Make our Planet Great Again" program 779 (Ref. ANR-17-MPGA-0017) for support. G.G. and G.T. would 780 also like to thank the Italian MIUR through the PRIN 781 2017Project Multi-e (20179337R7) "Multielectron transfer for 782 the conversion of small molecules: an enabling technology for 783 the chemical use of renewable energy" for financial support to 784 this work. The silicon carbide material was supplied by SICAT 785 SARL (www.sicatcatalyst.com), and Dr. Ch. Pham is gratefully 786 acknowledged for helpful discussion. SEM analysis was carried 787 out at the joint SEM platform of the ICPEES-IPCMS, and T. 788 Romero (ICPEES) is gratefully acknowledged for performing 789 the experiments.

790 **REFERENCES**

791 (1) Götz, M.; Lefebvre, J.; Mörs, F.; McDaniel Koch, A.; Graf, F.; 792 Bajohr, S.; Reimert, T. Renewable Power-to-Gas: A technological and 793 economic review. Renewable Energy 2016, 85, 1371-1390.

- 794 (2) Centi, G.; Quadrelli, E. A.; Perathoner, S. Catalysis for CO₂ 795 conversion: a key technology for rapid introduction of renewable 796 energy in the value chain of chemical industries. Energy Environ. Sci. 797 **2013**, *6*, 1711–1731.
- 798 (3) Hashimoto, K.; Kumagai, N.; Izumiya, K.; Takano, H.; Kato, Z. 799 The Production of Renewable Energy in the Form of Methane Using

Electrolytic Hydrogen Generation. Energy, Sustainability and Society 800 2014, 4, 17. 801

(4) Jentsch, M.; Trost, T.; Sterner, M. Optimal Use of Power-to-Gas 802 Energy Storage Systems in an 85% Renewable Energy Scenario. 803 Energy Procedia 2014, 46, 254-261. 804

(5) Rönsch, S.; Schneider, J.; Matthischke, S.; Schlüter, M.; Götz, 805 M.; Lefebvre, J.; Prabhakaran, S. Review on methanation-from 806 fundamentals to current projects. Fuel 2016, 166, 276-296. 807

(6) Wang, W.; Wang, S.; Ma, X.; Gong, J. Recent advances in 808 catalytic hydrogenation of carbon dioxide. Chem. Soc. Rev. 2011, 40, 809 3703-3727. 810

(7) Stankiewicz, A.; Moulijn, J. A. Process Intensification. Ind. Eng. 811 Chem. Res. 2002, 41, 1920-1924. 812

(8) Timko, B. P.; Whitehead, K.; Gao, W.; Kohane, D. S.; 813 Farokhzad, O.; Anderson, D.; Langer, R. Advances in Drug Delivery. 814 Annu. Rev. Mater. Res. 2011, 41, 1-20. 815

(9) Norris, M. D.; Seidel, K.; Kirschning, A. Externally Induced Drug 816 Release Systems with Magnetic Nanoparticle Carriers: An Emerging 817 Field in Nanomedicine. Adv. Ther. 2019, 2, 1800092. 818

(10) Hedayatnasab, Z.; Abnisa, F.; Daud, W. M. A. W. Review on 819 magnetic nanoparticles for magnetic nanofluid hyperthermia 820 application. Mater. Des. 2017, 123, 174-196. 821

(11) Lozinskii, M. G. Industrial Applications of Induction Heating; 822 Pergamon: New York, NY, USA, 1969; p 690. 823

(12) Wang, W.; Duong-Viet, C.; Xu, Z.; Ba, H.; Tuci, G.; 824 Giambastiani, G.; Liu, Y.; Truong-Huu, T.; Nhut, J.-M.; Pham-Huu, 825 C. CO₂ Methanation Under Dynamic Operational Mode Using 826 Nickel Nanoparticles Decorated Carbon Felt (Ni/OCF) Combined 827 with Inductive Heating. Catal. Today 2020, 357, 214-220. 828

(13) Wang, W.; Duong-Viet, C.; Tuci, G.; Liu, Y.; Rossin, A.; 829 Luconi, L.; Nhut, J. M.; Nguyen-Dinh, L.; Giambastiani, G.; Pham- 830 Huu, C. Highly Nickel-Loaded y-Alumina Composites for a 831 Radiofrequency-Heated, Low-Temperature CO2 Methanation 832 Scheme. ChemSusChem 2020, 13, 5468-5479. 833

(14) Bordet, A.; Lacroix, L.-M.; Fazzini, P.-F.; Carrey, J.; Soulantica, 834 K.; Chaudret, B. Magnetically Induced Continuous CO₂ Hydro- 835 genation Using Composite Iron Carbide Nanoparticles of Exception- 836 ally High Heating Power. Angew. Chem., Int. Ed. 2016, 55, 15894- 837 15898. 838

(15) De Masi, D.; Asensio, J. M.; Fazzini, P. F.; Lacroix, L. M.; 839 Chaudret, B. Engineering Iron-Nickel Nanoparticles for Magnetically 840 Induced CO2 Methanation in Continuous Flow. Angew. Chem., Int. 841 *Ed.* **2020**, *59*, 6187–6191. 842

(16) Rivas-Murias, B.; Asensio, J. M.; Mille, N.; Rodríguez-González, 843 B.; Fazzini, P. F.; Carrey, J.; Chaudret, B.; Salgueiriño, V. Magnetically 844 Induced CO₂ Methanation Using Exchange-Coupled Spinel Ferrites 845 in Cuboctahedron-Shaped Nanocrystals. Angew. Chem., Int. Ed. 2020, 846 59, 15537-15542. 847

(17) Niether, C.; Faure, S.; Bordet, A.; Deseure, J.; Chatenet, M.; 848 Carrey, J.; Chaudret, B.; Rouet, A. Improved Water Electrolysis Using 849 Magnetic Heating of FeC-Ni Core-Shell Nanoparticles. Nat. Energy 850 2018, 3, 476-483. 851

(18) Díaz-Puerto, Z. J.; Raya-Barón, Á.; van Leeuwen, P. W. N. M.; 852 Asensio, J. M.; Chaudret, B. Determination of the surface temperature 853 of magnetically heated nanoparticles using a catalytic approach. 854 Nanoscale 2021, 13, 12438-12442. 855

(19) Riedinger, A.; Guardia, P.; Curcio, A.; Garcia, M. A.; Cingolani, 856 R.; Manna, L.; Pellegrino, T. Subnanometer local temperature 857 probing and remotely controlled drug release based on azo- 858 functionalized iron oxide nanoparticles. Nano Lett. 2013, 13, 2399- 859 2406. 860

(20) Meffre, A.; Mehdaoui, B.; Connord, V.; Carrey, J.; Fazzini, P. 861 F.; Lachaize, S.; Respaud, M.; Chaudret, B. Complex Nano-objects 862 Displaying Both Magnetic and Catalytic Properties: A Proof of 863 Concept for Magnetically Induced Heterogeneous Catalysis. Nano 864 Lett. 2015, 15, 3241-3248. 865

(21) Hartman, T.; Geitenbeek, R. G.; Whiting, G. T.; Weckhuysen, 866 B. M. Operando monitoring of temperature and active species at the 867 single catalyst particle level. Nat. Catal. 2019, 2, 986-996. 868

869 (22) Tuci, G.; Liu, Y.; Rossin, A.; Guo, X.; Pham, C.; Giambastiani,
870 G.; Pham-Huu, C. Porous Silicon Carbide (SiC): a Chance for
871 Improving Catalysts or just Another Active Phase Carrier? *Chem. Rev.*872 2021, 121, 10559–10665.

873 (23) Ersen, O.; Florea, I.; Hirlimann, C.; Pham-Huu, C. Exploring 874 nanomaterials with 3D electron microscopy. *Mater. Today* **2015**, *18*, 875 395–408.

876 (24) Florea, I.; Ersen, O.; Hirlimann, C.; Roiban, L.; Deneuve, A.; 877 Houllé, M.; Janowska, I.; Nguyen, P.; Pham, C.; Pham-Huu, C. 878 Analytical electron tomography mapping of the SiC pore oxidation at 879 the nanoscale. *Nanoscale* **2010**, *2*, 2668–2678.

880 (25) Nguyen, P.; Pham, C. Innovative porous SiC-based materials: 881 From nanoscopic understandings to tunable carriers serving catalytic 882 needs. *Appl. Catal., A* **2011**, *391*, 443–454.

883 (26) Toby, B. H.; Von Dreele, R. B. GSAS-II: the genesis of a 884 modern open-source all purpose crystallography software package. *J.* 885 *Appl. Crystallogr.* **2013**, *46*, 544–549.

886 (27) For a standardized procedure see on "Datasheet & Manuals" at 887 the following link: https://www.optris.global/optris-cslaser-lt (ac-888 cessed October, 2021).

889 (28) Cairns, R. W.; Ott, E. X-Ray Studies of the System Nickel-

890 Oxygen—Water. I. Nickelous Oxide and Hydroxide. J. Am. Chem. Soc. 891 **1933**, 55, 527–533.

892 (29) Jette, E. R.; Foote, F. Precision Determination of Lattice 893 Constants. J. Chem. Phys. **1935**, *3*, 605–616.

894 (30) Jenkins, R.; Snyder, R. L. Introduction to X-ray Powder 895 Diffractometry; John Wiley & Sons Inc., 1996; pp 89–91.

(31) Liu, Y.; Edouard, D.; Nguyen, L. D.; Begin, D.; Nguyen, P.;
Pham, C.; Pham-Huu, C. High Performance Structured Platelet MilliReactor Filled with Supported Cobalt Open Cell SiC Foam Catalyst
for the Fischer-Tropsch Synthesis. *Chem. Eng. J.* 2013, 222, 265–273.
(32) Nesbitt, H. W.; Legrand, D.; Bancroft, G. M. Interpretation of

901 Ni2p XPS spectra of Ni conductors and Ni insulators. *Phys. Chem.* 902 *Miner.* **2000**, 27, 357–366.

903 (33) Liu, B.; Liu, L. R.; Liu, X. J.; Liu, M. J.; Xiao, Y. S. Variation of 904 crystal structure in nickel nanoparticles filled in carbon nanotubes. 905 *Mater. Sci. Technol.* **2012**, *28*, 1345–1348.

906 (34) Lin, Y.-R.; Ho, C.-Y.; Chuang, W.-T.; Ku, C.-S.; Kai, J.-J. 907 Swelling of ion-irradiated 3C–SiC characterized by synchrotronra-908 diation based XRD and TEM. *J. Nucl. Mater.* **2014**, 455, 292–296.

909 (35) Pujula, M.; Sánchez-Rodríguez, D.; Lopez-Olmedo, J. P.; Farjas, 910 J.; Roura, P. Measuring thermal conductivity of powders with 911 differential scanning calorimetry. *J. Therm. Anal. Calorim.* **2016**, 912 *125*, 571–577.

913 (36) Chen, D.-X.; Pascu, O.; Roig, A.; Sanchez, A. Size analysis and 914 magnetic structure of nickel nanoparticles. *J. Magn. Magn.* 2010, 322, 915 3834–3840.

916 (37) He, X.; Zhong, W.; Au, C.-T.; Du, Y. Size dependence of the 917 magnetic properties of Ni nanoparticles prepared by thermal 918 decomposition method. *Nanoscale Res. Lett.* **2013**, *8*, 446.

919 (38) Li, L.; Zheng, J.; Liu, Y.; Wang, W.; Huang, Q.; Chu, W.
920 Impacts of SiC Carrier and Nickel Precursor of NiLa/support
921 Catalysts for CO₂ Selective Hydrogenation to Synthetic Natural Gas
922 (SNG). ChemistrySelect 2017, 2, 3750–3757.

923 (39) Vrijburg, W. L.; Moioli, E.; Chen, W.; Zhang, M.; Terlingen, B.
924 J. P.; Zijlstra, B.; Filot, I. A. W.; Züttel, A.; Pidko, E. A.; Hensen, E. J.
925 M. Efficient base-metal NiMn/TiO₂ catalyst for CO₂ methanation.
926 ACS Catal. 2019, 9, 7823–7839.

(40) Xu, L.; Wang, F.; Chen, M.; Nie, D.; Lian, X.; Lu, Z.; Chen, H.;
Zhang, K.; Ge, P. CO₂ methanation over rare earth doped Ni based
mesoporous catalysts with intensified low-temperature activity. *Int. J.*Hydrogen Energy 2017, 42, 15523-15539.

931 (41) Asensio, J. M.; Miguel, A. B.; Fazzini, P. F.; van Leeuwen, P. W.
932 N. M.; Chaudret, B. Hydrodeoxygenation Using Magnetic Induction:
933 High-Temperature Heterogeneous Catalysis in Solution. *Angew.*934 Chem., Int. Ed. 2019, 58, 11306-11310.

935 (42) Niether, C.; Faure, S.; Bordet, A.; Deseure, J.; Chatenet, M.; 936 Carrey, J.; Chaudret, B.; Rouet, A. Improved water electrolysis using magnetic heating of FeC-Ni core-shell nanoparticles. *Nat. Energy* 937 2018, 3, 476. 938

(43) Périgo, E. A.; Hemery, G.; Sandre, O.; Ortega, D.; Garaio, E.; 939 Plazaola, F.; Teran, F. J. Fundamentals and advances in magnetic 940 hyperthermia. *Appl. Phys. Rev.* **2015**, *2*, 041302. 941

(44) Julian, I.; Ramirez, H.; Hueso, J. L.; Mallada, R.; Santamaria, J. 942 Non-oxidative methane conversion in microwave-assisted structured 943 reactors. *Chem. Eng. J.* **2019**, 377, 119764. 944

(45) Ramirez, A.; Hueso, J. L.; Mallada, R.; Santamaria, J. In situ 945 temperature measurements in microwave-heated gas-solid catalytic 946 systems. Detection of hot spots and solid-fluid temperature gradients 947 in the ethylene epoxidation reaction. *Chem. Eng. J.* **2017**, *316*, 50–60. 948

(46) Pearce, J.; Giustini, A.; Stigliano, R.; Jack Hoopes, P. Magnetic 949 Heating of Nanoparticles: The Importance of Particle Clustering to 950 Achieve Therapeutic Temperatures. *J. Nanotechnol. Eng. Med.* **2013**, *4*, 951 0110071–01100714. 952

(47) Moses, A. J. Eddy current losses in soft magnetic materials. 953 Wiley Encyclopedia of Electrical and Electronics Engineering; Wiley, 954 2016; pp 1–22. 955

(48) Appino, C.; de la Barrière, O.; Fiorillo, F.; Lobue, M.; 956 Mazaleyrat, F.; Ragusa, C. Classical eddy current losses in Soft 957 Magnetic Composites. J. Appl. Phys. **2013**, 113, 17A322–117A3223. 958

(49) Ruta, S.; Chantrell, R.; Hovorka, O. Unified model of 959 hyperthermia via hysteresis heating in systems of interacting magnetic 960 nanoparticles. *Sci. Rep.* **2015**, *5*, 9090. 961

(50) Hergt, R.; Dutz, S.; Röder, M. Effects of size distribution on 962 hysteresis losses of magnetic nanoparticles for hyperthermia. J. Phys.: 963 Condens. Matter 2008, 20, 385214. 964

(51) Kuz'min, M. D. Shape of Temperature Dependence of 965 Spontaneous Magnetization of Ferromagnets: Quantitative Analysis. 966 *Phys. Rev. Lett.* **2005**, 94, 107204. 967

(52) Crangle, J.; Goodman, G. M. The Magnetization of pure iron 968 and nickel. *Proc. R. Soc. London, Ser. A* **1971**, 321, 477. 969

(53) Carrey, J.; Mehdaoui, B.; Respaud, M. Simple Models for 970 Dynamic Hysteresis Loop Calculations of Magnetic Single-Domain 971 Nanoparticles: Application to Magnetic Hyperthermia Optimization. 972 J. Appl. Phys. **2011**, 109, 083921. 973

(54) Houlding, T. K.; Rebrov, E. V. Application of Alternative 974 Energy Forms in Catalytic Reactor Engineering. *Green Process. Synth.* 975 **2012**, *1*, 19–31. 976

(55) Sechovský, V. Encyclopedia of Materials: Science and Technology; 977
Buschow, K. H. J., Cahn, R. W., Flemings, M. C., Ilschner, B., Kramer, 978
E. J., Mahajan, S., Veyssière, P., Eds.; Elsevier: Amsterdam, 2001; p 979
5021. 980

(56) Frederikse, H. P. R. Handbook of Chemistry and Physics; 981 Properties of Magnetic Materials; CRC Press LLC: Boca Raton, FL, 982 2009. 983