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## Optimization of H<sub>2</sub>O<sub>2</sub> production in small-scale off-grid buffer layer flow cell equipped with Cobalt@N-Doped Graphitic Carbon Core–Shell Nanohybrid electrocatalyst

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### Abstract:

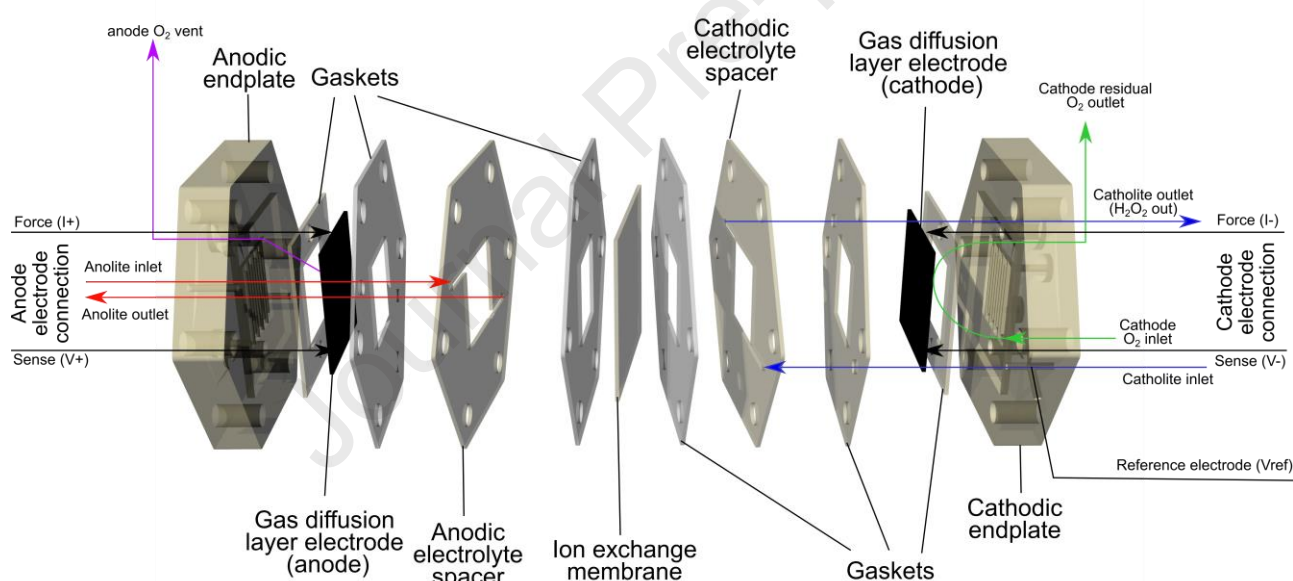
Electrochemical oxygen reduction (ORR) to hydrogen peroxide (H<sub>2</sub>O<sub>2</sub>) is emerging as a sustainable approach for the production of “green” H<sub>2</sub>O<sub>2</sub> requiring only oxygen and electricity compared to the energy intensive anthraquinone process. High 2e<sup>-</sup> selectivity is required in order to boost faradaic and energy efficiency (FE) of the process. Upon correct tuning of their properties, nitrogen-doped carbon materials are excellent candidates as electrocatalyst for H<sub>2</sub>O<sub>2</sub> electrosynthesis due to their chemical and electrochemical resistance and 2e<sup>-</sup> selectivity. Furthermore, careful cell design and parameter optimization are mandatory for an industrial scale up of the process. In this study, a Cobalt@N-doped graphitic carbon core–shell nanohybrid (CS(Co)-N-GC) electrocatalyst was studied in a buffer layer complete cell equipped with a proton exchange membrane in order to determine the effect of flow rate and potential on process selectivity and energy efficiency. After optimization, the cell was able to produce 0.5 wt% H<sub>2</sub>O<sub>2</sub> with an average FE higher than 40%, an energy consumption lower than 8 kWh kg<sub>H<sub>2</sub>O<sub>2</sub></sub><sup>-1</sup> and a production rate of 1.2 g h<sup>-1</sup> g<sub>cat</sub> @ 0.3V vs RHE with the possibility to produce up to 1 wt% H<sub>2</sub>O<sub>2</sub>.

### Introduction:

Hydrogen peroxide can be considered an excellent “green” oxidant due to the absence of byproducts (except water) upon use, relatively high redox potential, and relative safety and low toxicity. Its uses range from organic compound oxidation, (paper pulp) bleaching, detergent, chemical synthesis, textile industry, waste water treatment and medical uses (disinfection, sanitization, cleansing).[1] Currently, nearly all hydrogen peroxide worldwide (6 million metric tons) is produced by the energy intensive Anthraquinone process, that involves several steps of

reaction and separation followed by distillation.[1,2] To be economically sustainable the anthraquinone process requires very large scales and centralized production that also pose several safety issues due the high concentration needed for efficient transportation (above 60%) requiring distillation prior to shipping. Therefore, this also has a very strong impact on the total energy consumption of the  $\text{H}_2\text{O}_2$  production process, ranging between 12 and 17 kWh  $\text{kg}_{\text{H}_2\text{O}_2}^{-1}$ . [2] In addition, many final uses require relatively low concentrations: for example, medical uses where the concentration is often less than 3% for example, [3-7]  $\text{H}_2\text{O}_2$  can be used effectively for the disinfection of surfaces and ambient from Sars-Cov-2 virus with a concentration of 0.5 wt%. Decentralizing the production of  $\text{H}_2\text{O}_2$  can promote effectively the access to cheap disinfectants in remote areas aiding the fight against the pandemic. One interesting decentralized  $\text{H}_2\text{O}_2$  production approach is the electrosynthesis via oxygen reduction reaction (ORR) requiring, in principle, only oxygen (air), water and electricity.[8-10] In order to produce  $\text{H}_2\text{O}_2$ , the reaction should proceed through the 2 electron pathway, as any spurious 4 electron process reduces severely the current efficiency (and hence energy efficiency) of the entire process.[8,11-12] Hydrogen peroxide stability is also an issue. Many metal nanoparticle based electrocatalysts demonstrate interesting performance and efficiency both in half cells and in full cells, [8,13-17] but material supply limitations and cost and metal leaching phenomena can significantly limit the applicability, efficiency and durability of this technology. Materials development for ORR needs to take into careful consideration the differences in synthetic conditions, as small variations can lead to remarkably different catalytic behaviors. Encapsulated transition metals into carbon nanostructures have been reported with various metals such as Fe and Co and various nanocarbon morphology, exhibiting profound activity and selectivity differences in ORR.[18-21], In addition, the use of heteroatom-doped carbon-based metal free electrocatalysts[17-38] can overcome many of the limitations posed by metal nanoparticle electrocatalysts, such as leaching and passivation. Another possible approach is to embed the metal in a nitrogen-doped graphitic core-shell structure, in which the electronic levels of the carbon and the nitrogen are affected by the presence of the core metal,[18] which can be favorably tuned to boost two electron ORR selectivity.[39,44]. In addition, some nitrogen functionalities such as graphitic, pyrrolic, pyridinic are known to promote the 2e<sup>-</sup> oxygen reduction reaction mechanism.[38,43] In a previous work, a Cobalt@N-doped graphitic carbon core-shell nanohybrid electrocatalyst (CS(Co)-N-GC) has been shown to demonstrate high  $\text{H}_2\text{O}_2$  selectivity (up to 98% faradaic efficiency, FE) in half cells and at low overpotential.[44] Departing from ideal model studies, performed with techniques such as RRDE (rotating ring-disk electrode), it is important to consider that practical complete cell performance is also affected by  $\text{O}_2$  and  $\text{H}_2\text{O}_2$  transport phenomena (in particular  $\text{H}_2\text{O}_2$  residence times) and reaction media (interfering ions and pH) affecting  $\text{H}_2\text{O}_2$  stability.[45,46] The use of a GDL can effectively promote  $\text{O}_2$  transport at the cathode without  $\text{O}_2$  electrolyte solubility limitations.[39-42,46-48] Another important consideration is that forced convection (RRDE) experiments tend to overestimate the  $\text{H}_2\text{O}_2$  selectivity compared to practical systems:[12,45]  $\text{H}_2\text{O}_2$  concentration buildup, although highly desirable, can severely affect the reaction selectivity since it translates in an increase of  $\text{H}_2\text{O}_2$  concentration at the electrode interface, that can result in its further reduction to  $\text{H}_2\text{O}$  or non-electrochemical disproportionation to  $\text{O}_2$  and water. Chuan et al. [49] described a GDL-based complete cell able to withstand up to 20%  $\text{H}_2\text{O}_2$  by using a porous

layer separator with a membrane on each side (acidic at the anode, alkaline at the cathode) designed to recombine  $\text{HO}_2^-$  produced at the cathode with  $\text{H}^+$  (from  $\text{H}_2$  oxidation reaction) produced at the anode without contact of the resulting neutral  $\text{H}_2\text{O}_2$  with the electrodes. This very promising approach still depends on the availability of gaseous  $\text{H}_2$  for the anode, limiting its applicability in remote areas. Tuning the residence times and potentials can control effectively parasitic phenomena limiting the FE and increasing the energy requirements of the process.[12] Summarizing, most studies are conducted in half cells to have a first scrutiny of the performance, with limited but relevant investigations on full devices.[49] The present study provides a contribution on the cell engineering in order to minimize the performance losses with respect to the bench tests, and it is a key step for the implementation of real devices. Therefore, we describe the optimization of a two compartment gas diffusion layer flow cell equipped with a **CS(Co)-N-GC** electrocatalyst. Results indicate the possibility to produce build up a concentration of in solution up to 0.5 wt%  $\text{H}_2\text{O}_2$  with an average faradaic efficiency higher than 40% and up to 1 wt% with an average faradaic efficiency higher than 20%, with an energy demand lower than that of the anthraquinone process when limiting the concentration to 0.8 wt%  $\text{H}_2\text{O}_2$ .



**Figure 1:** double buffer layer, gas diffusion  $\text{H}_2\text{O}_2$  electrosynthesis cell structure

## 2 Methods:

### 2.1 Synthesis of the electrocatalysts

#### 2.1.1 Synthesis of Cobalt@N-Doped graphitic carbon core-shell nanohybrid electrocatalyst electrocatalyst (CS(Co)-N-GC).

The **CS(Co)-N-GC** nanomaterial was synthesized as described in [44]. The synthesis of the electrocatalyst was performed by thermal decomposition. A 1:1 solution of  $\text{Co}(\text{ac})_2 \cdot 4\text{H}_2\text{O}$  (Merck ACS reagent,  $\geq 98.0\%$ ) and a nitrogen-bearing organic precursor (imidazole, Merck ACS reagent

≥99.0%) was prepared and agitated overnight. The solvent was then removed, and the as-obtained solid was transferred into an open alumina ceramic vessel and treated in a tubular furnace under Ar flow in order to remove oxygen. After 1.5 h at 40°C under Ar atmosphere, the temperature was raised to 900°C with a ramp of 5°C min<sup>-1</sup> and hold for 2 h before cooling of the vessel to room temperature, the obtained pyrolyzed material was then ground by using an agate mortar to obtain a fine powder. It was then dispersed in 0.5M HClO<sub>4</sub> (Merck) and subsequently stirred at 80 °C for 4 h in order to remove eventual exposed metal impurities. The acid-treated sample was filtered and washed with water. The procedure was repeated twice, and then the obtained black powder was dried under vacuum and recovered.

### 2.1.2 Cathode preparation on carbon cloth

20mg of **CS(Co)-N-GC** were dispersed ultrasonically for 60 minutes in mixture of 1.5 g 2-propanol (Merck ACS reagent, ≥99.5%) and 0.5g of milliQ water, together with 65mg of a 5 wt% nafion oligomer solution in lower aliphatic alcohols (Merck). The uniform ink obtained was casted by subsequent deposition-drying cycles on a 37x37mm carbon cloth support (GPP050M from Cetech Co. Ltd) previously cleaned ultrasonically and tape-masked to expose an area of 6.25 cm<sup>2</sup> (figure S1). **CS(Co)-N-GC** loading (1.6-1.9 mg cm<sup>-2</sup>) was estimated by weight difference.

### 2.1.3 IrO<sub>2</sub> anode preparation on carbon cloth

20mg of powdered IrO<sub>2</sub> electrocatalyst mixed in a mortar with 25mg of Vulcan XC-72 carbon (Cabot) were dispersed ultrasonically for 60 minutes in mixture of 1.5 mg 2-propanol and 0.5 g of water, together with 96 mg of a 5 wt% nafion oligomer solution in lower aliphatic alcohols. The uniform ink obtained was casted by subsequent deposition-drying cycles on a 37x37mm carbon cloth support (GPP050M from Cetech Co. Ltd) previously cleaned ultrasonically and tape-masked to expose an area of 6.25 cm<sup>2</sup> (see scheme S1) until the maximum possible loading on the surface was achieved. The IrO<sub>2</sub> loading (3 mg cm<sup>-2</sup>) was estimated by weight difference.

## 2.2 Electrochemical reactor experiments

Electrochemical reactor experiments were conducted in a 3D-printed ASA (acrylonitrile styrene acrylate) gas diffusion flow reactor (figure 1, S2, S3) equipped with two flow buffer layers (cathode and anode) with a circulating 0.1N H<sub>2</sub>SO<sub>4</sub> (Sigma Pure PA) solution in milliQ water on both compartments. Both the two liquid circuits (anode and cathode) were circulated by a Gilson peristaltic pump equipped with Tygon™ tubes. The cell, vertically oriented has four compartments: starting from cathode side, i) a gas compartment where pure O<sub>2</sub> (Air-liquide 99.95% purity) was flowed continuously at atmospheric pressure; the gas diffusion electrode with the oxygen reduction reaction electrocatalist (CS(Co)-N-GC) was housed at the interface between the gas compartment and the cathode buffer layer; liquid and gas tightness was ensured by two hexagonal rubber gaskets with the additional purpose of masking the active area (25 x 25 mm, 6.25 cm<sup>2</sup> exposed surface; ii) and iii) cathodic and anodic buffer layers flowing through the channels left by the spacers between the electrodes and the Nafion 117 membrane; and iv) the anode gas compartment where water oxidation reaction (WOR) takes place. The anode buffer layer and the anode gas compartment were separated by the carbon cloth GDL covered with IrO<sub>2</sub>

WOR electrocatalyst. The cathode feeding channel housed also the reference electrode (Ag|AgCl|KCl<sub>sat</sub>) in close proximity to the cathode electrocatalyst to minimize the uncompensated resistance. The buffer layers were filled with circulating 0.05M H<sub>2</sub>SO<sub>4</sub> (Merck Pure PA) and O<sub>2</sub> was fed continuously to the cathode gas compartment to ensure proper mass transport. Two different experiment types were performed: i) single pass experiments where the electrolyte, enriched with synthesized H<sub>2</sub>O<sub>2</sub> by oxygen reduction reaction, was not recycled and was directly collected into the final tank and ii) recirculation experiments, where the electrolyte was continuously recirculated into the cathodic chamber during the experiment.

The flow rate were adjusted from 0.28 and 13.7 mL min<sup>-1</sup> for buffer layers and the cathode potential was set in a range from 0.5 V to 0.1 V vs RHE. The total cell potential was measured by an external voltmeter in order to calculate the energy consumption of H<sub>2</sub>O<sub>2</sub> production. The current collectors were made of stainless steel in a control-sense double connector configuration (figure S2) to minimize the contact IR<sub>drop</sub>. The solution was not allowed to come in contact with the steel to avoid Fe ions contamination inside the cell that would decompose the H<sub>2</sub>O<sub>2</sub>, lowering the efficiency. The collected H<sub>2</sub>O<sub>2</sub> was sampled accordingly with the optimal concentration range of the analytical method chosen for the determination. H<sub>2</sub>O<sub>2</sub> for tolerance experiments was purchased from Merck (35%, no stabilizers) and titrated prior to the experiment directly sampling from the cell before applying potential.

Air breathing proof-of-concept experiments were performed with the gas diffusion layers directly exposed to the room atmosphere by using open chamber and plates both on cathode and anode side.

The temperature was held constant at 25°C by circulating the electrolyte into a 50cm silicone coiled tube immersed into a thermostatic bath.

### 2.3 Product analysis

The collected sample solutions H<sub>2</sub>O<sub>2</sub> concentrations and quantity were determined by permanganometric titration using a 2-2.5 mM potassium permanganate solution as titrant and indicator freshly prepared and standardized prior to each experiment using high purity sodium oxalate (dried at 80°C for 24 hours prior the preparation of the standard solution) as mother substance. The more concentrated H<sub>2</sub>O<sub>2</sub> solutions were sampled by using an 50-1000ul Eppendorf automatic pipette and scaled accordingly. The faradaic efficiency was determined by comparing charge passed with the charge corresponding to the moles of H<sub>2</sub>O<sub>2</sub> detected by using Equation 1. The anode compartment was analyzed in order to detect any H<sub>2</sub>O<sub>2</sub> formation or membrane crossover that resulted in quantities lower or close to the detection limit of the technique and negligible compared to the total charge passed (less than 1% of the cathodic H<sub>2</sub>O<sub>2</sub> detected)

$$FE = 5F \frac{V_{liq} M_{MnO_4} V_{eq}}{V_{smp} Q_{tot}}$$

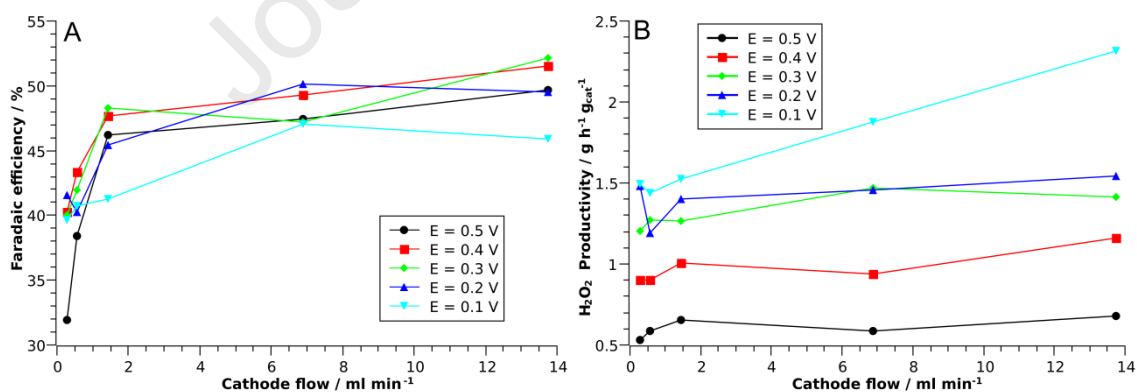
**Equation 1:** faradaic efficiency (FE); M<sub>MnO<sub>4</sub></sub> is the molarity of the titrant; V<sub>eq</sub> is the endpoint volume, Q<sub>tot</sub> is the total charge passed in the experiment; V<sub>liq</sub> is the total liquid sample volume; V<sub>smp</sub> is the volume of the sample effectively titrated; F is Faraday's constant 96 485 A mol<sup>-1</sup>.

### 2.3 Electrocatalyst durability analysis

In order to determine any morphological changes to the electrocatalyst over a long period of  $\text{H}_2\text{O}_2$  electrosynthesis, after the long  $\text{H}_2\text{O}_2$  production experiment, the cathode of the flow cell was recovered, washed with deionized water, dried under a current of  $\text{N}_2$ , and then the electrocatalyst powder was removed from the carbon cloth by mechanical scraping. The powder was then analyzed with Transmission Electron Microscopy (TEM) using a JEOL 2200FS microscope working at 200 kV, equipped with a High-Angle Annular Dark-Field (HAADF) detector, an in-column omega filter; the microanalysis was performed with an Energy-Dispersive X-ray (EDX) spectrometer. The pristine electrocatalyst powder morphological and compositional was carried out using a Tescan GAIA 3 FIB/SEM microscope equipped with an Edax EDS Octane detector. Both Images and EDS compositional characterization were acquired using an impinging beam energy of 10 keV. Element distribution maps were acquired with a collection time of 600s.

## 3 Results and Discussion

The first set of experiments aimed at determining the optimum conditions for the electrocatalyst and the setup, in particular applied cathode potential and electrolyte flow rate. The first experiments at low flow rate were performed in “single pass” mode, without electrolyte recycling, in order to determine the effect of the fluid velocity; then the flow rate was further increased and the mode was switched to “recycle”. The data obtained for the optimization of flow rate are shown in figure 2. The effect of the flow rate on energy efficiency is shown in the electronic SI (figure S4).



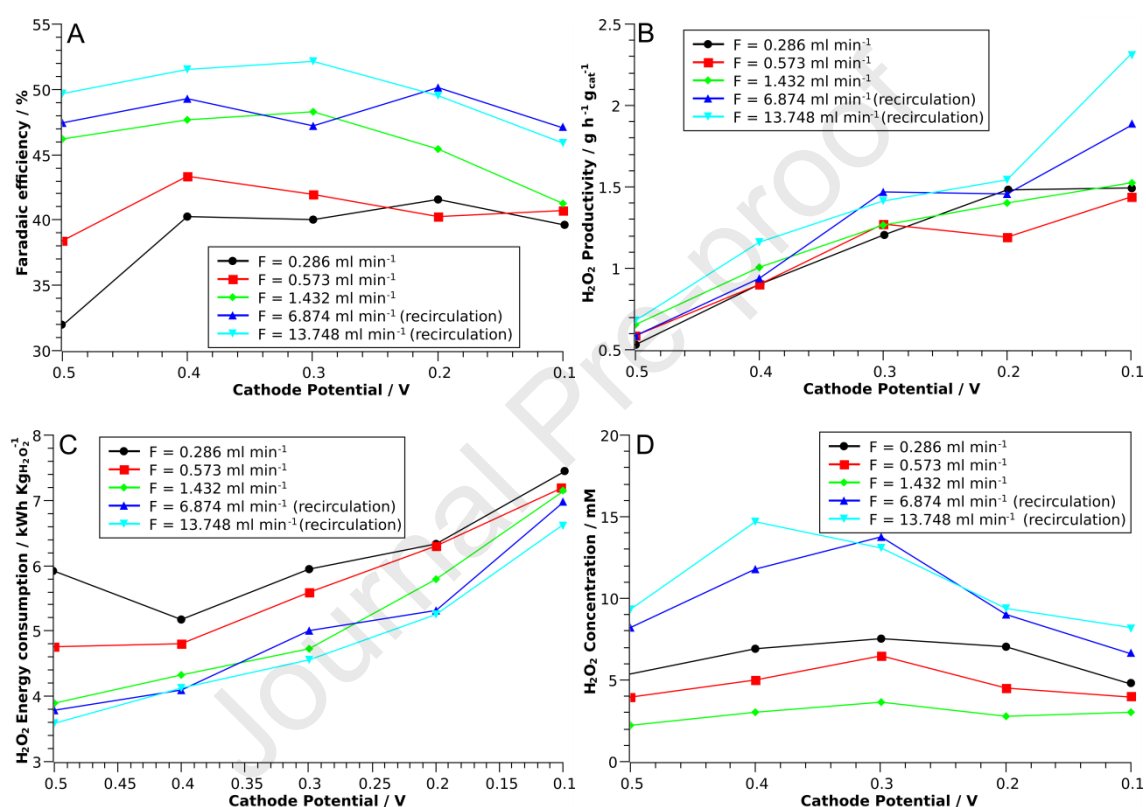
**Figure 2:** Cathode flow rate effect of faradaic efficiency (a) and on productivity (b) of the cell equipped with the CS(Co)-N-GC GDL cathode. The data points at 7 and 14 ml min<sup>-1</sup> were performed in recirculation mode.

A higher flow rate was beneficial for the faradaic efficiency ( $\text{FE}_{\text{H}_2\text{O}_2}$ ) as it reduces residence time and reduces the diffusion boundary layer thickness, speeding the mass transport and decreasing the cathode interface  $\text{H}_2\text{O}_2$  concentration preventing its further reduction to  $\text{H}_2\text{O}$ . As stated



previously, forced convection experiments do not take into account  $\text{H}_2\text{O}_2$  accumulation phenomena, due to fast mass transport and low bulk  $\text{H}_2\text{O}_2$  concentration.

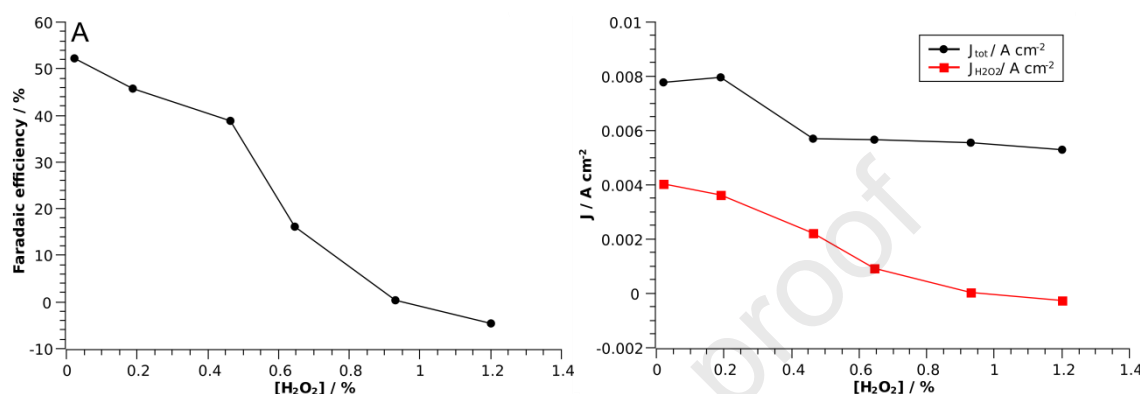
Regarding the effect of potential (figure 3), the more cathodic the potential was, the higher the current and the productivity; the FE increased till 0.3V vs RHE then started declining probably due to a promotion of 4 electron pathway rather than product accumulation, as the drop in selectivity seems more severe in stronger convection experiments ruling out a problem of  $\text{H}_2\text{O}_2$  accumulation at the electrode interface. The peak FE was 52% @ 0.3V at a flow rate of  $13.8 \text{ ml min}^{-1}$ . The energy consumption per unit of  $\text{H}_2\text{O}_2$  increased accordingly with the decline of faradaic efficiency and higher  $\text{IR}_{\text{drop}}$  in the cell and electrolyte remaining, though, remaining under  $10 \text{ kWh kg}_{\text{H}_2\text{O}_2}^{-1}$ .



**Figure 3:** Cathode potential effect of FE (a), productivity (b), energy consumption (c) and  $\text{H}_2\text{O}_2$  final concentration (d) of the cell equipped with the CS(Co)-N-GC GDL cathode. The points at 7 and 14  $\text{ml min}^{-1}$  were performed in recirculation mode.

In order to survey the effect of hydrogen peroxide bulk concentration ( $[\text{H}_2\text{O}_2]_{\text{bulk}}$ ) in the electrolyte on the faradaic efficiency, a series of experiments of  $\text{H}_2\text{O}_2$  electrosynthesis in the presence of increasing starting hydrogen peroxide concentrations were performed (figure 4). Those experiments were designed to determine the impact of  $[\text{H}_2\text{O}_2]$  on the selectivity excluding from the picture the electrocatalyst degradation that may occur in very long experiments. The conditions chosen for the experiments were 0.3V RHE and  $13.8 \text{ ml min}^{-1}$  flow rate. The faradaic efficiency starts at 52% declining to 40% at 0.5 wt%  $\text{H}_2\text{O}_2$ , then rapidly declining to zero (production of  $\text{H}_2\text{O}_2$  equals its degradation) at 0.9 wt%, getting negative (-4% FE) at 1.2 wt% indicating that  $\text{H}_2\text{O}_2$  (which is added at the start of the experiment) is consumed by

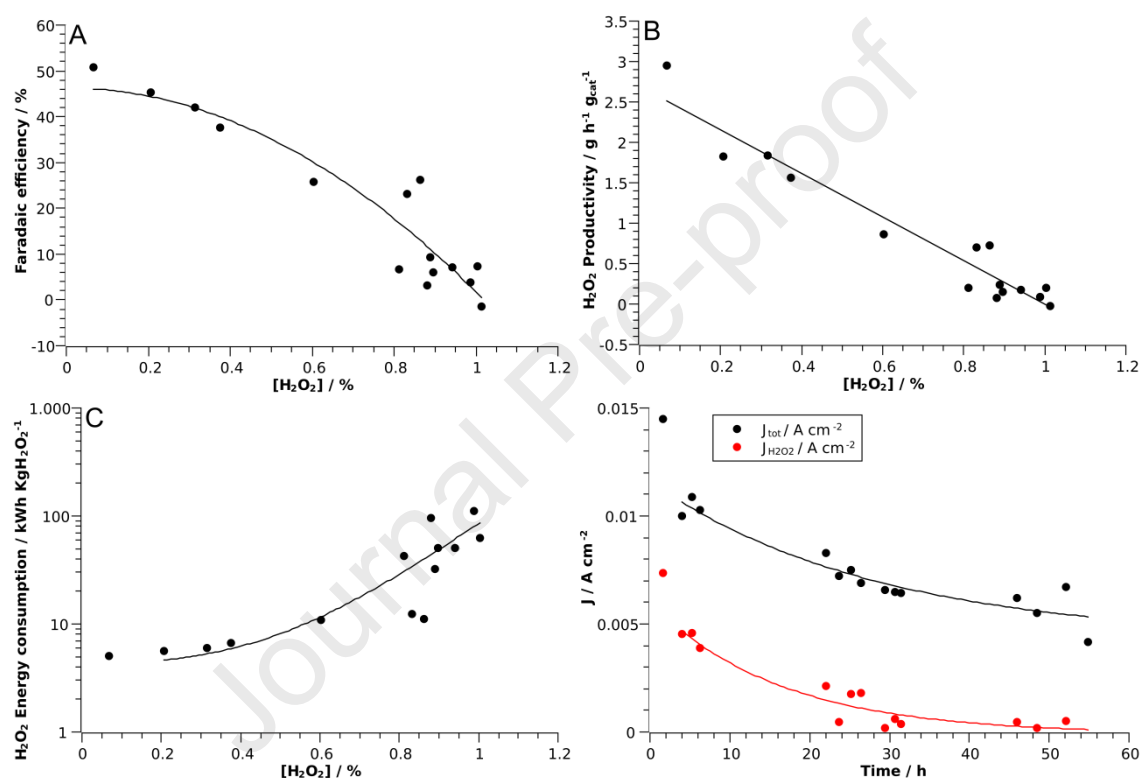
decomposition rather than being solely produced at the cathode. The  $\text{H}_2\text{O}_2$  partial current density decreased accordingly while the total current remained constant beyond 0.5 wt%. The maximum concentration attainable with this electrocatalyst in real conditions reported around 0.9 wt%. A long experiment (figure 5 and S5-S8), starting from zero  $[\text{H}_2\text{O}_2]_{\text{bulk}}$ , was also performed under the same conditions (0.3V vs RHE,  $13.8 \text{ ml min}^{-1}$ ,  $25^\circ\text{C}$ ). The  $\text{FE}_{\text{H}_2\text{O}_2}$  (figure S5) starts around 52% declining to 25% in 20 hours and to 0% in 55 hours. The  $\text{H}_2\text{O}_2$  bulk concentration increases (figure S6) up to 1% where it reaches a plateau corresponding to 0%  $\text{FE}_{\text{H}_2\text{O}_2}$ .



**Figure 4:** Faradaic efficiency (a) and on current densities (b) as a function of initial  $[\text{H}_2\text{O}_2]$  for the complete cell equipped with the CS(Co)-N-GC GDL cathode at the optimal flow and potential condition.

Notably the  $\text{FE}_{\text{H}_2\text{O}_2}$  profile with respect to the concentration (figure 5a) declines rather linearly with the  $[\text{H}_2\text{O}_2]_{\text{bulk}}$  reaching zero at 1 wt%. The same declining trend was observed with partial current and hence productivity starting from  $3 \text{ g h}^{-1} \text{ g}_{\text{cat}}^{-1}$ . Interestingly, also the total current decreased suggesting that the  $\text{H}_2\text{O}_2$  degradation could not be attributed to further reduction to  $\text{H}_2\text{O}$  but to a non-electrochemical disproportionation reaction at the interface. Replacing the solution with fresh electrolyte ( $[\text{H}_2\text{O}_2]_{t=0} = 0$ ) restored completely the  $\text{FE}_{\text{H}_2\text{O}_2}$  to 50% ruling out electrocatalyst degradation. For this reason, the maximum experiment time was limited to 50 hours due to the accumulation of  $\text{H}_2\text{O}_2$  leading to the decrease of FE but a subsequent cycle produced nearly the same FE efficiency profile but a higher overall cell voltage due to the anode degradation. It is important to point out that very long operation times (total time > 2000 hours) request a careful design of membrane and anode to avoid their degradation, leading eventually to an increase in cell voltage and hence the decrease in total energy efficiency; in particular OER (oxygen evolution reaction) is very difficult in acidic media as corrosion of anode electrocatalyst is severe. Regarding the cathodic side, the end-of-cycle *ex-situ* HAADF-STEM images, EDX elemental maps and High Resolution TEM (HRTEM) micrographs (figures 6, S12-S14) show the conservation of the core-shell structure of the electrocatalyst, with the absence of cobalt leaching phenomena outside the graphitic structure. In addition, surface SEM-EDAX mapping of pristine powder shows homogeneous distribution (figure S15-S17, table S1) of nitrogen and cobalt in the sample. Furthermore, analyzing the anode feed the total  $\text{H}_2\text{O}_2$  detected accounted for less than 1% of the total charge passed, ruling out massive cross-over losses as the reason of such a decline in  $\text{FE}_{\text{H}_2\text{O}_2}$ . The energy consumption follows the same trend of the FE increasing massively beyond 0.6 wt%  $[\text{H}_2\text{O}_2]_{\text{bulk}}$ . Below this concentration the process outperforms anthraquinone process in terms of

energy demand [2], with the possibility of producing cheaper disinfectant-grade  $\text{H}_2\text{O}_2$  on demand locally, requiring only water, air (oxygen) and electricity, that can be used for water purification, surface cleansing and disinfection uses. The reactor can be designed to limit the concentration below the 0.5 wt% threshold by continuously replacing (or diluting) the electrolyte circulating into the cell. The system can be easily scaled up for higher production rates. A  $1 \text{ m}^2$  electrode assembly can produce continuously up to  $4 \text{ liters h}^{-1}$  of  $\text{H}_2\text{O}_2$  0.5 wt% solution ( $20 \text{g H}_2\text{O}_2$ ) consuming less than  $160 \text{W}$  in the process (up to  $32 \text{ liters daily}$  in an  $8 \text{ hour}$  time using a  $160 \text{W}$  solar panel). Employing more  $\text{H}_2\text{O}_2$ -tolerant electrocatalysts and further improving mass transport (for example introducing vortex generators inside the electrolyte buffer layer) could increase the energy efficiency and the production rate even further.



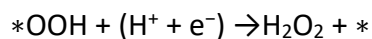
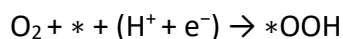
**Figure 5:**  $[\text{H}_2\text{O}_2]$  effect on faradaic efficiency (a), productivity (b) and energy consumption (c), partial and total current density time profile (d) during the 55 hours long  $\text{H}_2\text{O}_2$  generation experiment performed with the cell equipped with the CS(Co)-N-GC GDL cathode.

We have reported a literature survey with some cases of cells (table 1) reporting, except one case with a peculiar cell configuration,[49] low maximum  $\text{H}_2\text{O}_2$  concentrations attainable; this is due to the permanence of the  $\text{H}_2\text{O}_2$  at the electrode leading to further reduction or disproportionation drastically reducing its FE.

A proof-of-concept experiment of an air-breathing system was also performed (figure S9-S11). The performance is lower than the setup using pure oxygen and the attainable concentration is roughly 25% of the experiment performed in pure oxygen, however this setup uses only water ( $1 \text{ mol}$  per mol of  $\text{H}_2\text{O}_2$ ), oxygen from air ( $1/2 \text{ mol}$  per mol of  $\text{H}_2\text{O}_2$ ) and electricity without the need to supply pure  $\text{O}_2$  from a cylinder. Considering the same  $10 \text{ kWh kg}_{\text{H}_2\text{O}_2}^{-1}$  energy consumption

threshold to evaluate the performance, this setup can attain a concentration of 0.11 wt% H<sub>2</sub>O<sub>2</sub> before the energy consumption becomes higher than that of the anthraquinone process.[2]

The mechanism of oxygen reduction to H<sub>2</sub>O<sub>2</sub> has been extensively studied and follows the two step pathway:[37,38,43,45,46]

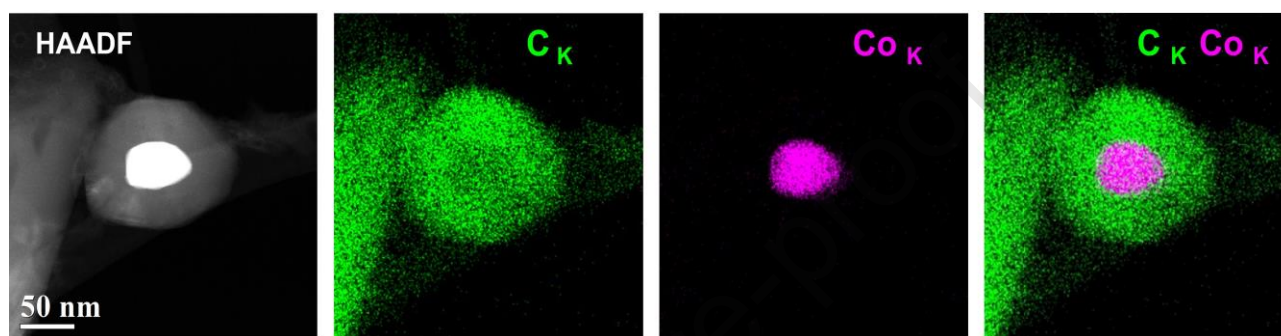


Where \* represent an adsorption site.

We have previously shown how the selectivity towards H<sub>2</sub>O<sub>2</sub> by N-doped carbon catalysts is strongly affected by the type and distribution of N moieties within the carbon as graphitic, pyrrolic, pyridinic are known to promote the 2e<sup>-</sup> oxygen reduction reaction mechanism.[36,38,43,50] Moreover, the porosity and textural properties of the materials play an important role due to their effect on the mass transport, so that here the catalyst's structure is tailored to maximize efficiency in H<sub>2</sub>O<sub>2</sub> production taking into account all these issues which we had been previously explored.[44]

Ref	Average Faradaic Efficiency	Maximum H <sub>2</sub> O <sub>2</sub> concentration	Energy consumption kWh kg <sub>H<sub>2</sub>O<sub>2</sub></sub> <sup>-1</sup>	Type of experiment	Average Current density
This paper	40%	0.5%	<8	BFL-O <sub>2</sub>	8 mA cm <sup>-2</sup>
	20%	0.11%	<10	BFL-Air	4 mA cm <sup>-2</sup>
23	70-95%	0.2-0.75%	-	H-cell	4-12 mA cm <sup>-2</sup>
39	>85	-	-	H-cell	0.15 mA cm <sup>-2</sup>
44	>95%	-	-	H-cell	0.9 mA cm <sup>-2</sup>
47	50-85%	-	4.6-19.4	BFL-Air	25-200 mA cm <sup>-2</sup>
49	90%	20%	4.5 (including H <sub>2</sub> )	H <sub>2</sub> -anode fed, double membrane buffer layer fuel cell	300 mA cm <sup>-2</sup>
51	50%	0.05%	9-11	BFL-O <sub>2</sub>	50 mA cm <sup>-2</sup>

**Table 1:** Literature overview of some H<sub>2</sub>O<sub>2</sub> oxygen-reduction-based electrosynthesis experiments; BFL-O<sub>2</sub> = Buffer layer cell (pure O<sub>2</sub> fed); BFL-Air = Buffer layer cell (air breathing); H-cell = membrane-separated H-cell



**Figure 6:** HAADF-STEM image and EDX elemental maps of the electrocatalyst after the long H<sub>2</sub>O<sub>2</sub> electrosynthesis experiment, showing the conservation of the core-shell electrocatalyst structure.

#### 4 Conclusions

In conclusion, a Cobalt@N-Doped graphitic carbon core-shell nanohybrid ORR electrocatalyst (CS(Co)-N-GC) operating in a complete gas-fed O<sub>2</sub> buffer layer electrochemical reactor was studied to determine the effect of flow rate and potential on process selectivity and energy efficiency. After optimization, the cell was able to produce continuously 0.5 wt% H<sub>2</sub>O<sub>2</sub> with negligible electrocatalyst degradation attaining an average faradaic efficiency for H<sub>2</sub>O<sub>2</sub> higher than 40%, consuming less than 8 kWh kgH<sub>2</sub>O<sub>2</sub><sup>-1</sup> and demonstrating a production rate of 1.2 g h<sup>-1</sup> g<sub>cat</sub> @ 0.3V vs RHE corresponding to 4 liters of 0.5 wt% H<sub>2</sub>O<sub>2</sub> each hour with an average energy consumption of 0.16 kWh and requiring only air, water and electricity. This will allow access to cheap and environmentally friendly disinfectants in remote areas, without the massive environmental footprint of centralized H<sub>2</sub>O<sub>2</sub> production processes and their logistics. For future developments, the total energy balance must be also considered and careful design of the other cell parts (anode and membrane) are mandatory for a long term operation. In this work, the energy balance was limited to the electrical energy cost (excluding pumps and heating) as the experiment was designed as a table top lab scale proof of concept with laboratory apparatus (such as peristaltic pumps) with no long term cell operation assessment due to the scarce IrO<sub>2</sub> anode stability in the operation conditions. In, forthcoming studies, a full scale up of the system and an LCA analysis will allow to determine the total energy cost of H<sub>2</sub>O<sub>2</sub> production through oxygen reduction reaction, in order to fathom for possible real device developments.

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The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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