

Fig. 4. Crystal structures of C₂ and C₃ highlighting the oxygen and H₂ sublattices. (*A* and *B*) Histograms of the number of neighboring O–O and H₂–H₂ as a function of distance in the C₂ and C₃ phase, respectively, at a pressure of 40 GPa. (*C*) Crystal structure of C₂. (*D* and *E*) Crystal structure of C₃. Oxygen atoms and H₂ centers of mass are shown as red and white spheres, respectively.

ice structures and account for the very high hydrogen density of this hydrate. While several gas hydrates were reported to have structural transitions at about 1–2 GPa (7, 8), hydrogen hydrate is only the second gas hydrate to show a phase transition above 10 GPa and to survive Megabar pressures. For methane hydrate, a structural transition from phase III (MH-III) to phase IV (MH-IV) is observed under compression to about 40 GPa at room temperature (36, 38). This is a transition between two different water sublattices for a constant methane-to-water molar ratio. Comparing the water frameworks, we can identify a common skeleton of all the filled ice structures showing remarkable high-pressure stability: the C_2 and C_3 phases and the methane hydrate phase MH-IV (36) share the ice I frame, either ice Ih or ice Ic, as shown in Fig. 5. The recurrence of the ice I network at high pressure is the result of an optimal compromise between the maximization of the hydrogen bond strength and a low efficiency-



Fig. 5. Comparison between the crystal structures of the (low-pressure) water ice phases *lh* and *lc* and those of the related (high-pressure) phases C_2 and C_3 of hydrogen hydrate, and MH-IV of methane hydrate. Oxygen and hydrogen atoms of the water molecules are shown as large red and small pink spheres in a ball-and-stick model. The centers of mass of H_2 and C_4 are shown as cyan and brown spheres, respectively. Hydrogen bonds are shown as black solid lines.

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packed water structure—about 1/3 for both ice I*h* and I*c*—which thus retains the ability to host small molecules.

The high-pressure phase of methane hydrate (MH-IV) has been shown to survive to pressure up to 150 GPa at least (36); however, it is presently unclear whether it is thermodynamically stable or only metastable (see for example ref. 39). The volume of methane hydrate is indeed higher than that of the sum of its separated components (water ice and pure solid methane), in both phases MH-III and MH-IV. The comparison between the volume of methane hydrate (38) and of the components (31, 40) is reported in SI Appendix, Fig. S10. On the other hand, the hydrogen hydrate phase C2 also has a larger volume than the sum of its separated pure components but C₃ is presently unique in being slightly denser than its constituent elements (Fig. 2C). According to our calculations, C3 remains thermodynamically stable against decomposition up to 200 GPa at least (SI Appendix, Fig. S1). By contrast, the hydrates of argon, krypton, xenon, and carbon dioxide have all been reported to fully decompose upon compression to pressures of about 1 to 6 GPa (41, 42), and for nitrogen hydrate, a recent work reported no stable solid phases from 5 GPa to 140 GPa (43). Yet, it seems unlikely that methane and hydrogen hydrates are exceptions and future studies will be able to verify whether other small non-polar guests, such as for example, helium, neon, or dioxygen can form a similar structure with water at extreme pressures. It is possible that an entire class of filled ice I gas hydrates exists at pressures compatible with those encountered in the interior of large gaseous and icy planets of our solar system or Neptune-like exoplanets.

Materials and Methods

Experimental Details. Starting polycrystalline hydrogen hydrate samples having the clathrate sll structure were prepared by exposing H₂O ice (Ih) to gas H₂ at 0.28 GPa for 20–30 min, following the method described in ref. 44. The ice was made of spheres with typical diameters of a few tens of micrometers. After preparation, the samples were recovered at ambient pressure and stored at liquid nitrogen temperature. For the high-pressure experiments, a small amount of sample was loaded at low temperature in a diamond anvil cell that was partially immersed in a liquid nitrogen bath. The loaded sample was then compressed to pressures of a few GPa before being warmed up to room temperature. This is a similar loading procedure to what we had previously done for methane hydrate (35, 36). This procedure inevitably produces a mixture of C₂ hydrogen hydrate and excess pure ice, because of the difference in molar ratios between the clathrate sII phase and the C2 phase. Clathrate sII samples prepared following our method have a molar ratio H₂O:H₂ between 4 and 5, as we have verified by neutron diffraction in the past (44), and the C₂ phase is characterised by a molar ratio of 1. One sample was prepared directly in the DAC by loading room-temperature liquid H₂O and high-pressure H₂ at 0.14 GPa using a gas loading setup. This sample was measured by Raman spectroscopy and by XRD. Similarly to the cryo-loaded samples, it also formed hydrogen hydrate with the structure C_2 and after laser heating at 40 GPa, with the structure C_3 . Some of the samples contained a small amount of pure N_2 and/or pure O_2 , both of which can be easily identified from their respective vibron Raman peaks. Those must have been trapped in the sample chamber during cryo-loading and acted as a pressure-transmitting medium. Some of the samples did not contain N₂ or O₂ and an example of Raman spectrum for such a sample is reported in SI Appendix, Fig. S11. Culet diameters ranging between 100 and 300 μ m for the diamond anvils were employed. Re-foil gaskets were used to contain the sample.

A Nd-YAG laser system (wavelength of 1.064 μ m) was employed to heat the sample, using gold as the laser coupler. Angle-dispersive X-ray powder diffraction patterns were acquired at ID15b at ESRF (Grenoble, France) using a monochromatic X-ray beam ($\lambda = 0.41$ Å) and an Eiger 2 9M CdTe flat-panel detector, with a typical acquisition time of a few seconds. The beam spot size was 6×6 μ m. The 2D diffraction patterns were treated to mask the Bragg peaks of the diamond anvils then integrated into one-dimensional patterns, and a smooth polynomial background was subtracted. Le Bail refinements were performed using FullProf. In the XRD measurements, pressure was determined using the equation of state of gold from ref. 45. Raman spectra were acquired using a commercial Horiba Jobin-Yvon LabRam HR800 Raman spectrometer in a backscattering geometry, equipped with a COBOLT SambaTM 1000-mW green laser. Acquisition time was typically of about a minute. A smooth background was subtracted from the Raman spectra before fitting the data and examples of fits for C₂ and C₃ at 50 and 75 GPa are reported in *SI Appendix*, Fig. S12. Reference spectra were recorded at each pressure from the gasket close to the sample chamber and particular care was taken to check the effect of the background subtraction on the fit results. In the Raman measurements, pressure was determined by the shift of the R1 ruby fluorescence line (46), from the edge of the stressed diamond signal (47) or from the vibron frequency of pure solid H₂ (34). All measurements were performed at room temperature.

Many sample loadings were performed for this study, as mentioned above, and Fig. 3 results from data merged together from different loadings. In SI Appendix, Fig. S13, we show Raman spectra and corresponding vibron frequencies for one single sample loading over the whole investigated pressure range, before and after laser heating. Several loaded samples were compressed and measured by Raman spectroscopy without performing laser heating, applying different compression rates to check whether the C₃ phase would form. Some indications of the possible formation of a small amount of C3 at room temperature above 70 GPa have been observed in some of the Raman experiments (*SI Appendix*, Fig. S14) but this point requests further investigation. One sample containing the C₂ phase was decompressed and measured by Raman spectroscopy. The corresponding frequencies are reported in panel A of SI *Appendix*, Fig. S15, where it can be seen that the pressure dependence of the peak frequencies is the same upon compression and decompression. Furthermore, we observe that no major changes occur upon decompression in the relative peak intensities of C_2 hydrogen hydrate and of pure H_2 , meaning that the partial decomposition of the hydrate is irreversible. We also decompressed the C₃ phase by Raman spectroscopy and observed it can be recovered down to at least 36 GPa at room temperature (SI Appendix, Fig. S15B). The stability of the C₃ phase under decompression is an important information in order to envisage the use of high-density hydrates as H₂ storage material and future studies will be able to check whether the C₃ phase can be recovered at ambient pressure and low temperature.

Simulation Methods. Crystal structure prediction was performed using evolutionary algorithm as implemented in the USPEX code (27, 28). The structures were relaxed using a five-step process with the Vienna ab initio simulation package (VASP) as a DFT code (48), employing the projector-augmented wave pseudopotentials with Perdew-Burke-Ernzerhof pseudopotentials provided with the package (49). Further details are provided in *SI Appendix*.

The C₂ and C₃ phases were simulated using Density Functional (Perturbation) Theory as implemented in Quantum ESPRESSO (50, 51), using Optimized Norm-Conserving Vanderbilt pseudopotentials (ONCV) (52) and optimized Becke88 van der Waals exchange-correlation functional (53-55). A cutoff of 90 Ry was employed on the plane waves' expansion, and the integration of the Brillouin zone was performed over a $4 \times 4 \times 4$ grid in reciprocal space with fixed occupations. The vibrational zero-point energy was computed within the harmonic approximation over a 2×2×2 grid in reciprocal space, and integrated over a 8×8×8 grid using Fourier interpolation. We did not include temperature effects in our calculations of the thermodynamics properties. Proton-ordered configurations for the water molecules satisfying the Bernal-Fowler rules were obtained directly from the USPEX structural search calculations. Furthermore, consideration of the H₂ molecules as two individual atoms lowers the crystal symmetry in the simulations: The simulated structures are not cubic but tetragonal or orthorhombic (see SI Appendix for further details). Details of our C2 and C₃ structures and unit cell dimensions as a function of pressure, calculated within DFT, are reported in SI Appendix, Table S2 and Fig. S8, respectively. For pure hydrogen and pure water, we used as references the (proton-ordered) structures of phase II (56) and ice VIII, respectively.

Anharmonicity and quantum nuclear effects were evaluated using the stochastic self-consistent harmonic approximation as implemented in the SSCHA code (29, 30). The SSCHA minimizes the free energy $F[\hat{\rho}] = \langle H \rangle_{\hat{\rho}} - TS[\hat{\rho}]$ by optimizing a trial density matrix $\hat{\rho}$ constrained among solutions of an auxiliary harmonic Hamiltonian. The free energy minimization is performed using the second-order Newton method, with a stochastic sampling of the Born-Oppenheimer energy landscape at DFT level (30). The method allowed us to evaluate the strength of quantum and anharmonic effects in the frequencies of the hydrogen vibron compared to the harmonic approximation. This method has been reported as the most successful to compute the phase diagram and Raman and IR spectra of H₂ vibrons at high pressure (57–60). The interested reader can find further information on the specific details employed in the simulation of C2 and C₃ in *SI Appendix* and additional details on the numerical implementation in ref. 29.

Data, Materials, and Software Availability. All study data are included in the article and/or *SI Appendix*.

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